



# Air Electrode Design and Preparation for Lithium Air Battery Application

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**And**

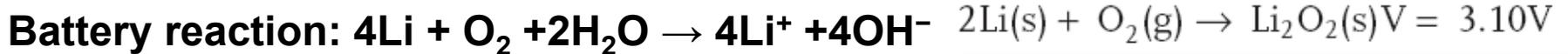
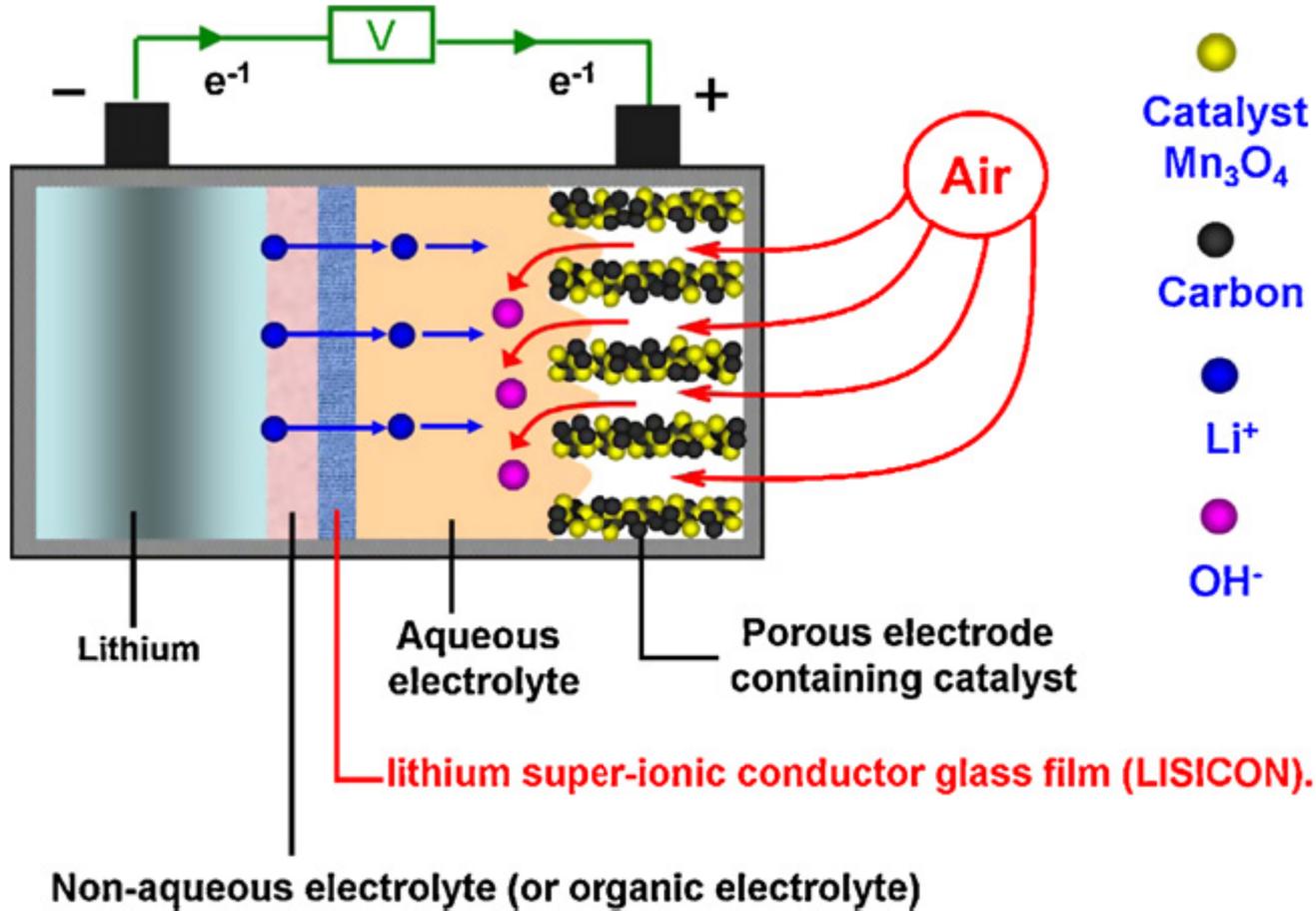
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**6th US-China Electric Vehicles and Battery Technology Workshop,  
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# Principle of Li-air battery





# Challenges of Li-air battery

**High** theoretical energy density is about **11972 Wh/kg**

**Low practical capacity**



**Low practical energy density**

**Low round trip efficiency**

- ◆ **Poor high-rate dischargeability**
- ◆ **Short cycle life** (Fast degradation under ambient conditions)
- ◆ **Safety issues**
- ◆ **Waterproof** (Isolated from water and oxygen permeating )
- ◆ **Unclear reaction mechanism at cathode electrode**



## Objectives of SJTU-UM Projects on Li-Air battery

- ◆ **Characterization of the catalyst microstructure and morphology during charge/discharge cycles**
- ◆ **Identification of the rate limiting mechanisms for decomposition of  $\text{Li}_2\text{O}_2$  (the primary discharge product) during recharge in order to understand the origin of the high charging potential**
- ◆ **Identification of mechanisms by which cathode catalysts facilitate discharge and recharge**
- ◆ **Based on identified mechanisms, screen for optimal catalyst compositions and morphologies**

## Deliverables

- ◆ **A comprehensive and computational model for creation and analysis of Li-air batteries**
- ◆ **Novel materials and structure for high capacity Li-air batteries**
- ◆ **Prototype of Li-air batteries**



## Critical Path

- ◆ Complete calculation of concentrations and diffusivities of all relevant neutral and charged intrinsic defects in  $\text{Li}_2\text{O}_2$  at fixed oxygen chemical potential and consistent with the “bulk” morphology of catalytic decomposition.
- ◆ Prepare various catalysts and evaluate their structure-related catalytic performance and characteristics; Identify impact of transition metal catalysts on defect formation energies through “extrinsic dopant” approach.
- ◆ Prepare prototype Li-air batteries with various components. Determine the effect of electrodes and electrolytes on the electrochemical performance.

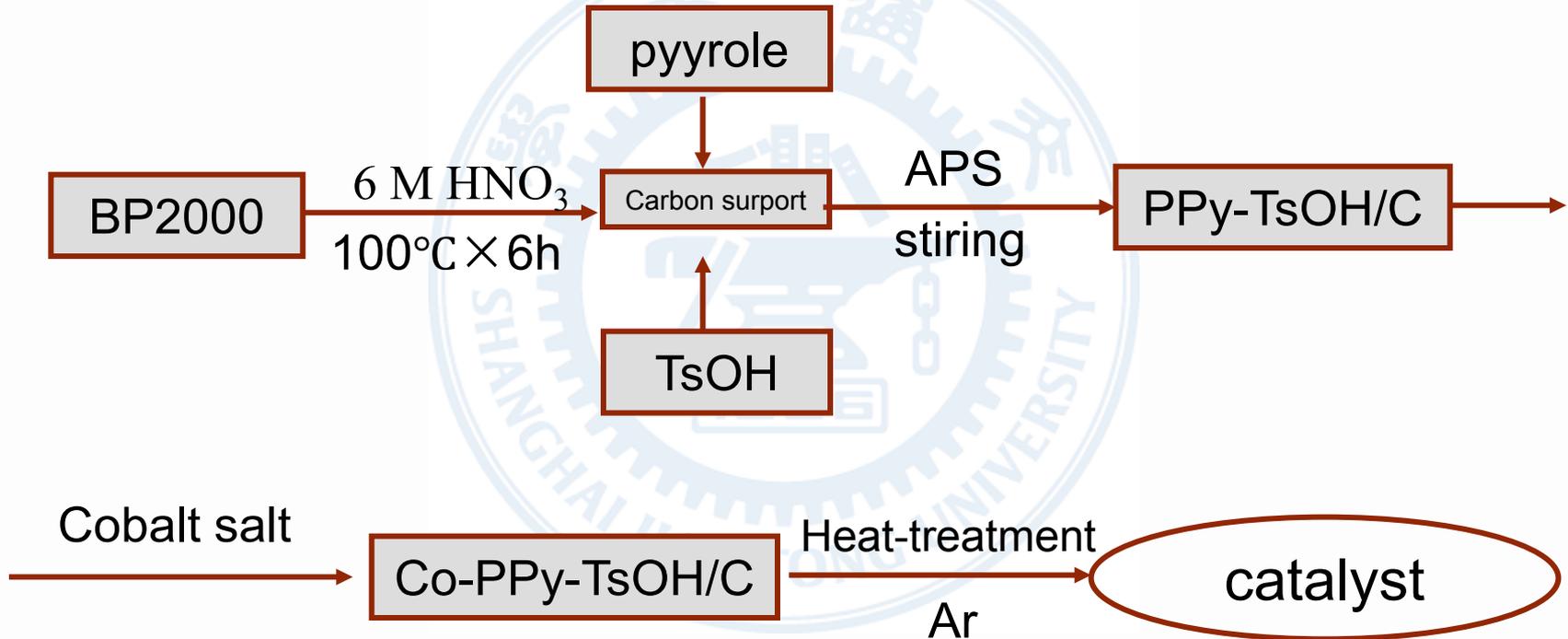


## Research progress at SJTU side

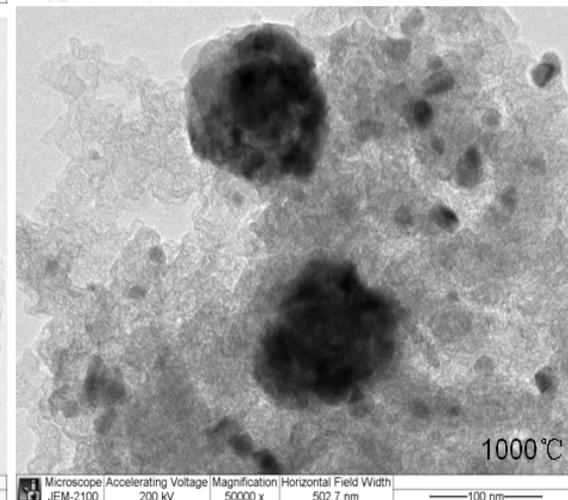
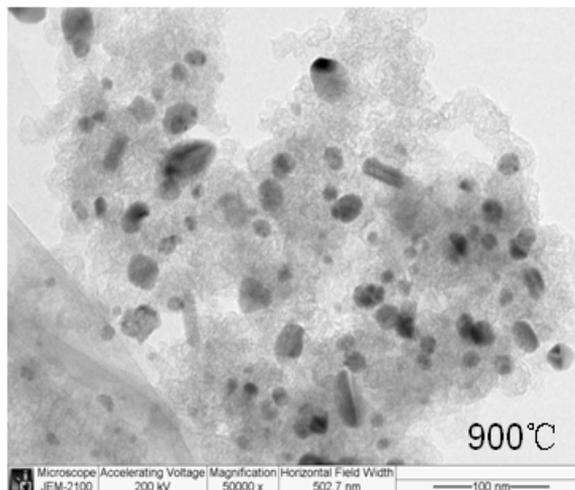
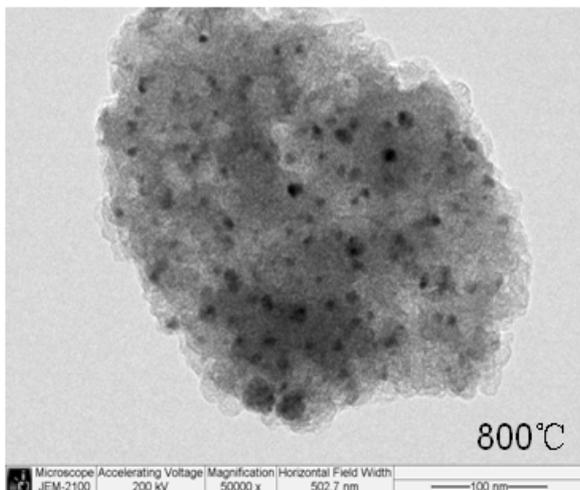
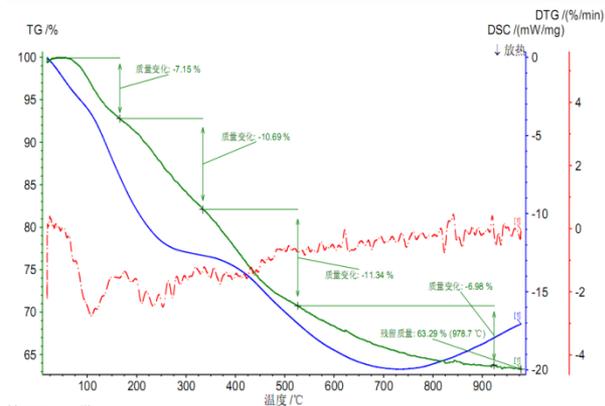
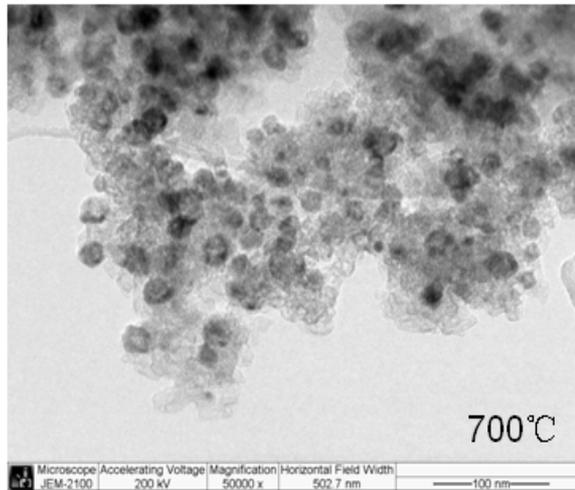
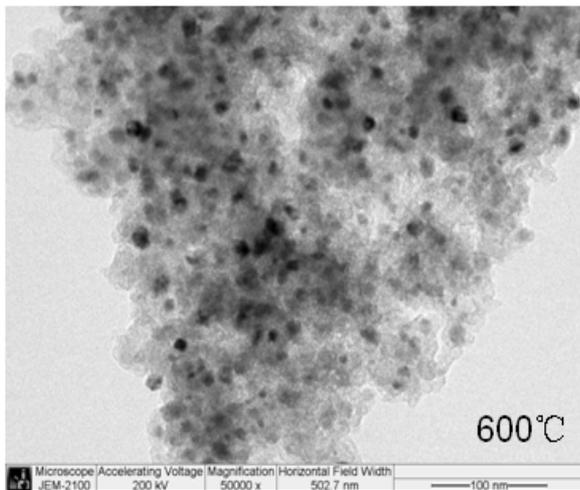
- ◆ Cathode catalyst design and development
  - ✓ Co-PPy-TsOH/C
  - ✓ Co-N/C
  - ✓ MnO<sub>x</sub> based catalysts
  - ✓ 1.0 M LiTFSI-DMMP based electrolyte
- ◆ Prototype Li-air battery design and development



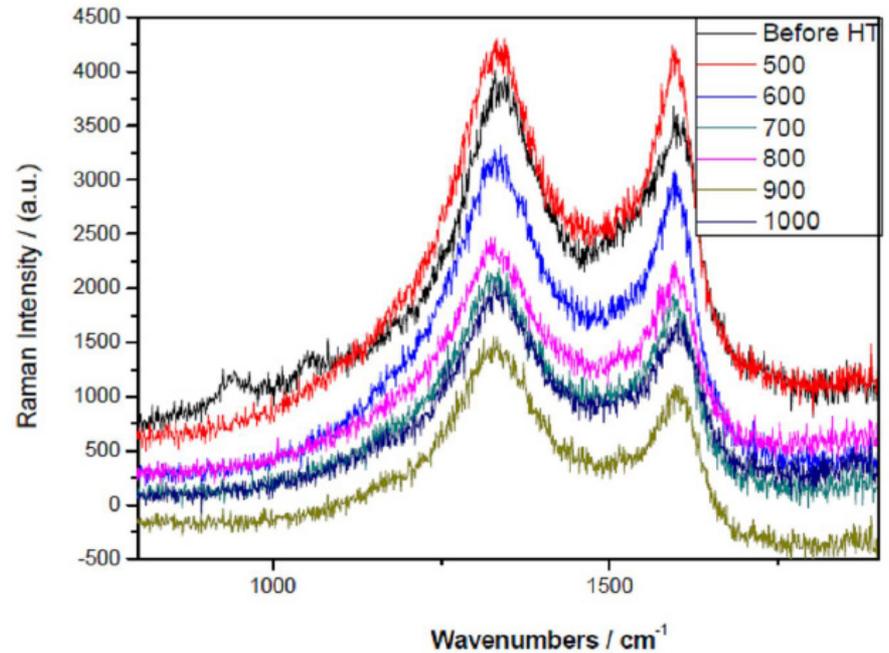
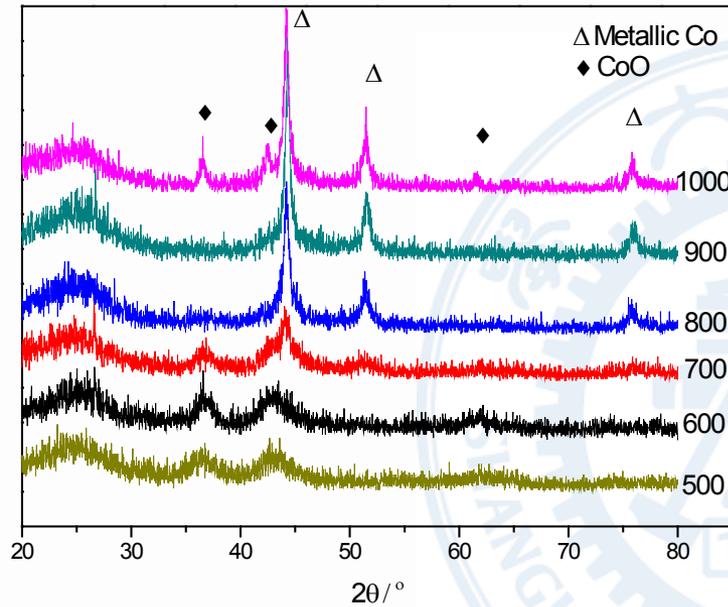
## Effect of heat-treatment conditions on catalytic performance of Co-PPy-TsOH/C toward oxygen reduction reaction



Xianxia Yuan, et al Improved Performance of Proton Exchange Membrane Fuel Cells with p-Toluenesulfonic Acid-Doped Co-PPy/C as Cathode Electrocatalyst. *Journal of the American Chemical Society*. 132(2010)6:1754- 1755



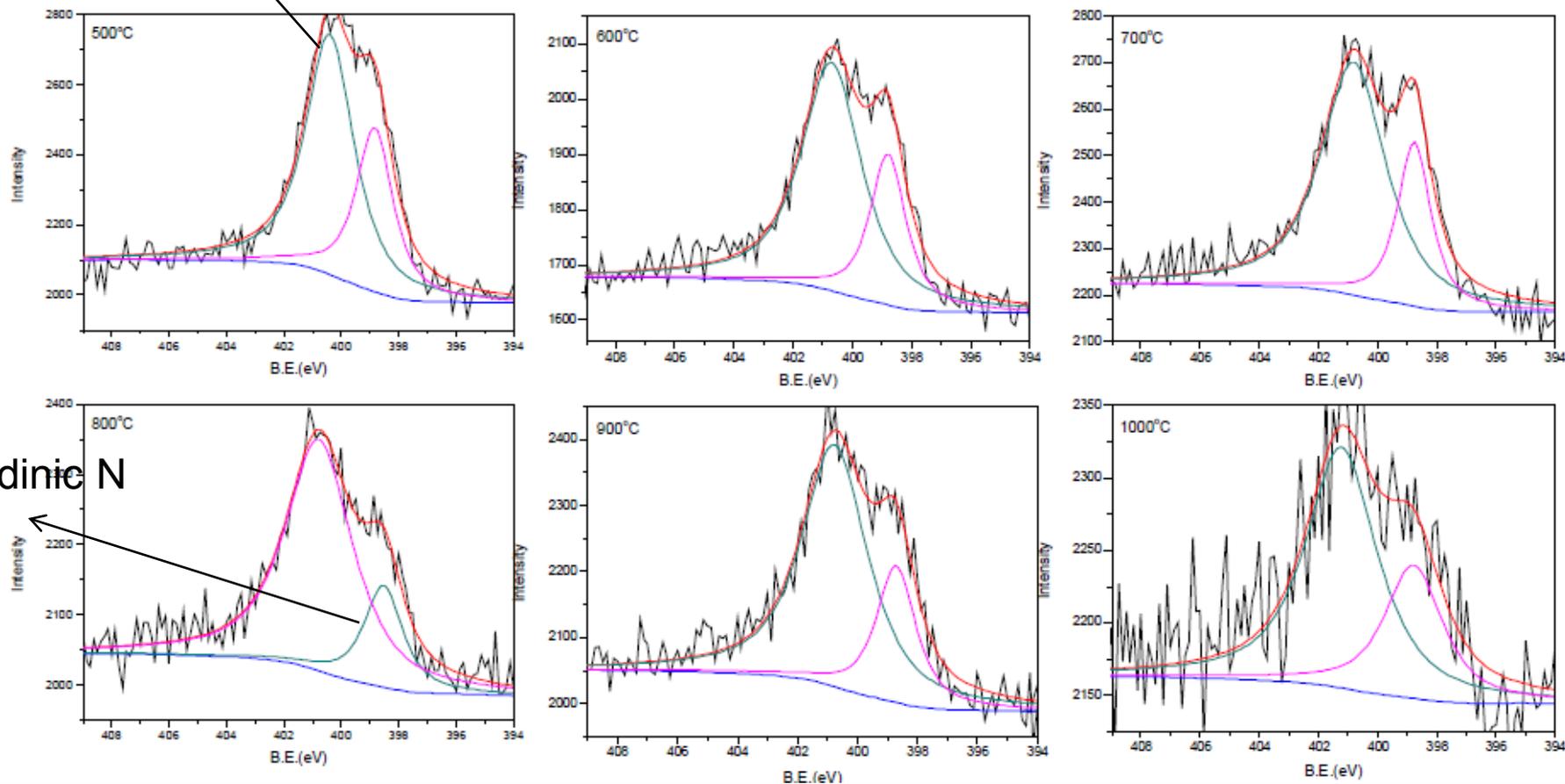
TEM images of Co-PPy-TsOH/C heat-treated at various temperatures



**XRD patterns and Raman spectra of Co-PPy-TsOH/C catalysts heat-treated at various temperatures**



pyrrolic N



pyridinic N

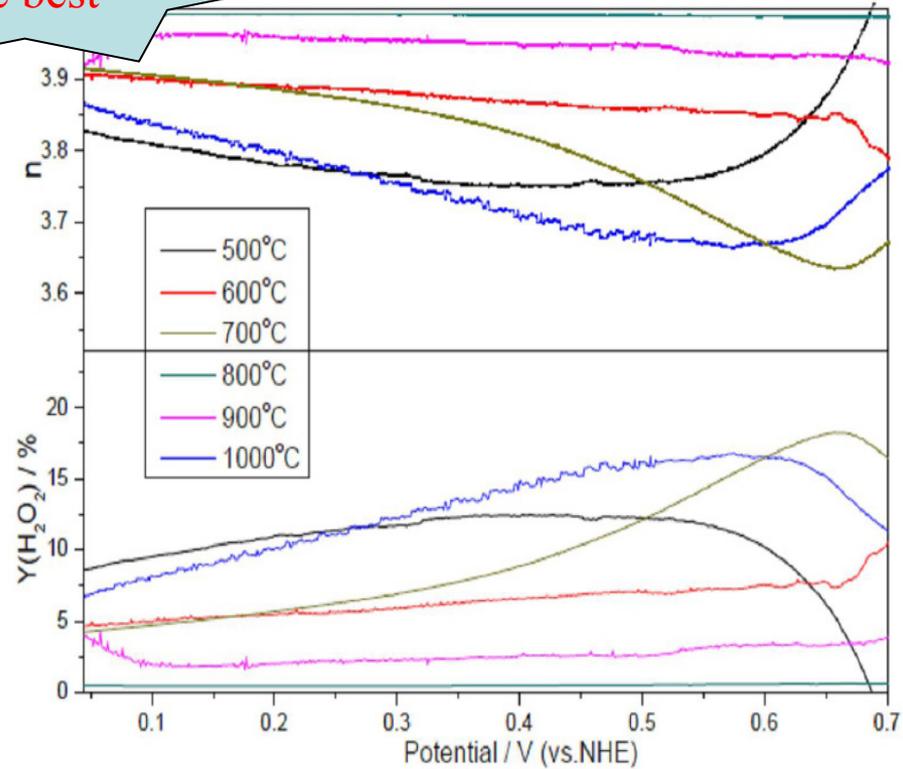
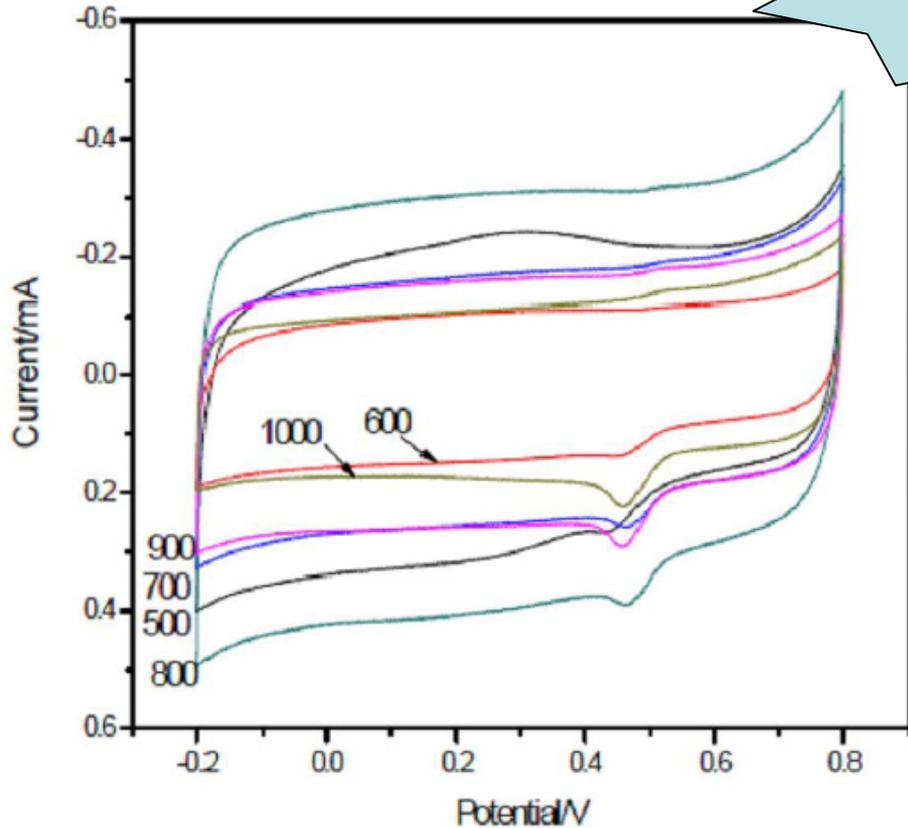
**XPS  $N_{1s}$  curves of catalyst Co-PPy-TsOH/C heat-treated at various temperatures**

## Area ratio of pyrrolic N/pyridinic N for of catalyst Co-PPy-TsOH/C heat-treated at various temperatures

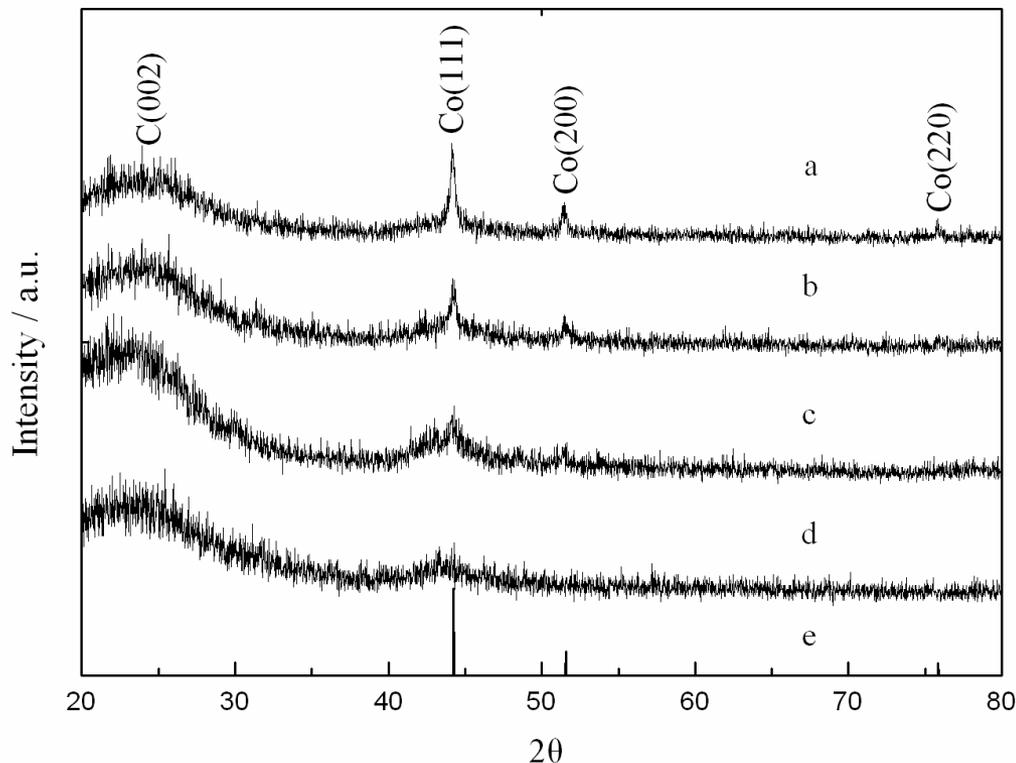
Heat treatment temperature / °C	Peak position of pyridinic N /ev	Peak position of pyrrolic N /ev	Area ratio of pyrrolic N/pyridinic N
500°C	398.837	400.399	1.9
600°C	398.792	400.706	2.5
700°C	398.761	400.814	2.7
800°C	398.526	400.814	3.9
900°C	398.702	400.759	2.9
1000°C	398.788	401.255	2.2



800 °C  
the best

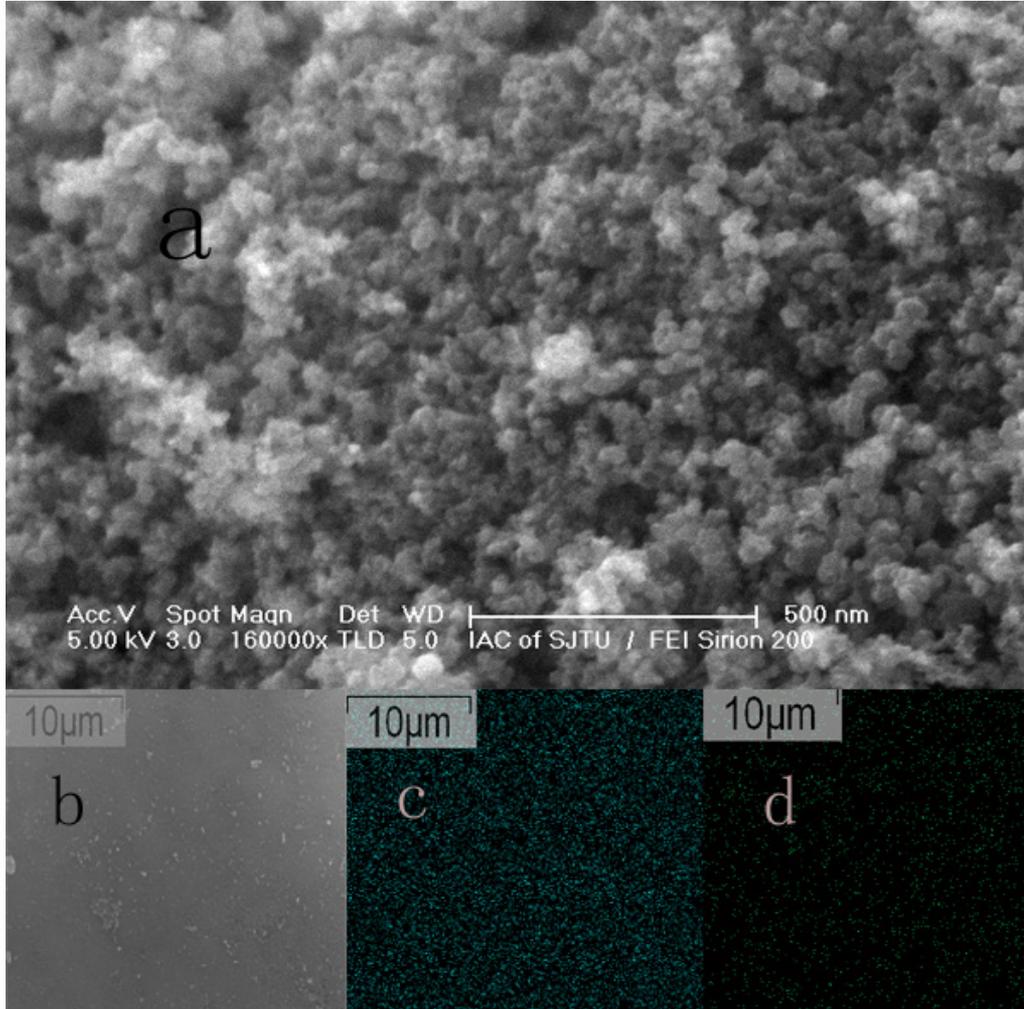


**Electro-catalytic performance of catalyst Co-PPy-TsOH/C heat-treated at various temperatures**



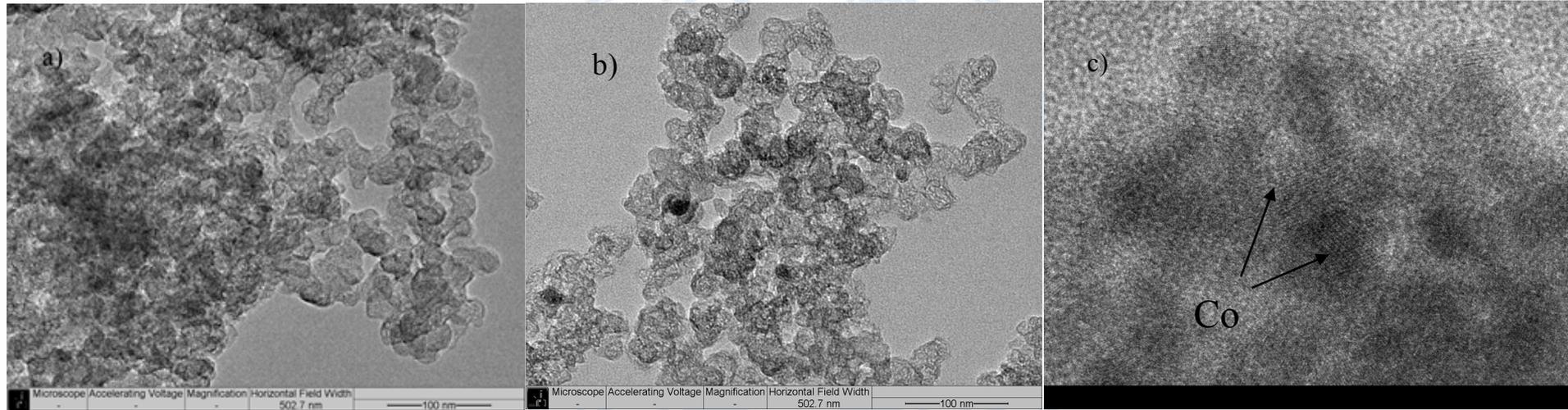
- a: 6%Co载量
- b: 3%Co载量
- c: 1%Co载量
- d: BP2000
- e: PDF89-4307

**A simple and cheap Phenanthroline (phen) was used as a ligand to prepare Co(phen)<sub>2</sub> complexes. The prepared Co(phen)<sub>2</sub> complexes were coated on BP2000 then heat-treated to obtain carbon-supported Co-N/C catalyst).**

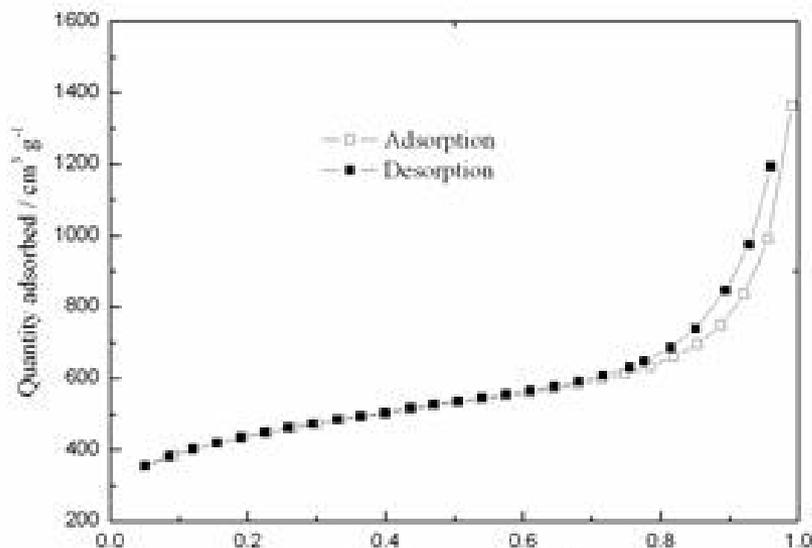
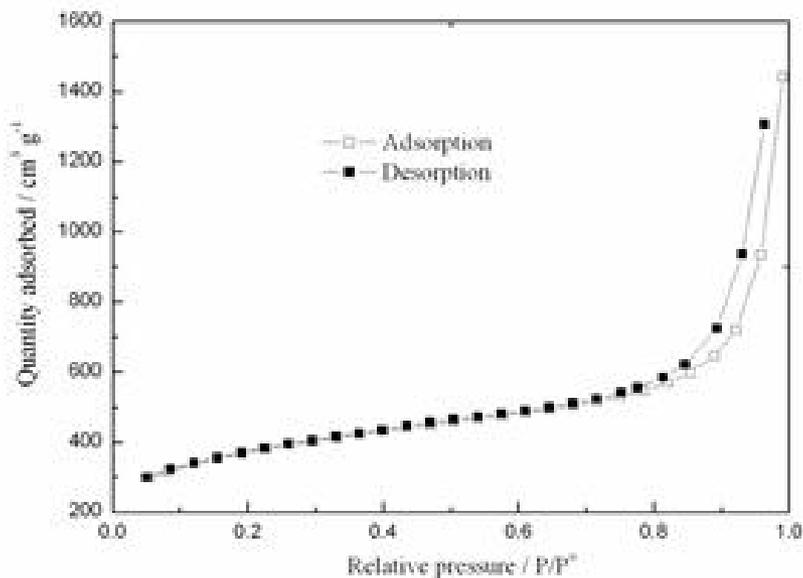


Most of the particles are distributed in the range of 20-40 nm. EDS mapping results corresponding to CoK $\alpha$  and NK $\alpha$  on Co-N show fairly uniform distribution of these elements, indicating that the Co-N composite in carbon support were well dispersed. It has the beneficial effect of promoting the reversibility of discharge product formation and decomposition.

C: Co elements  
D: N elements



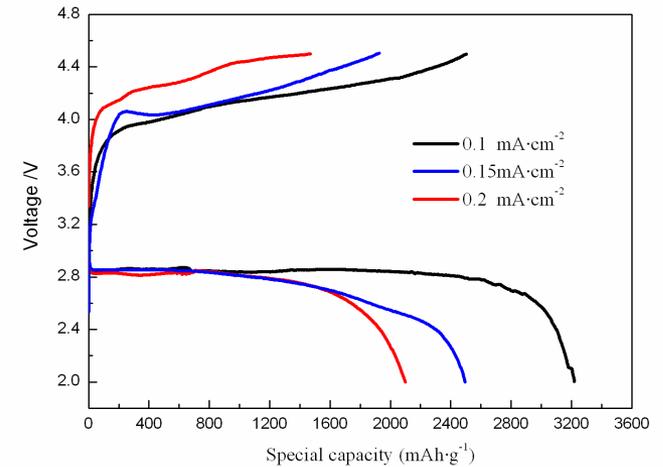
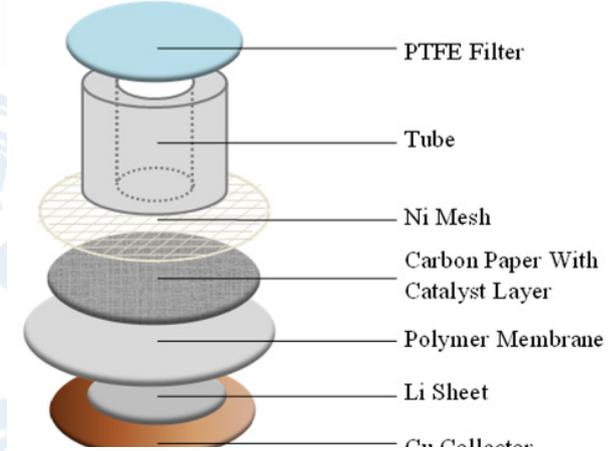
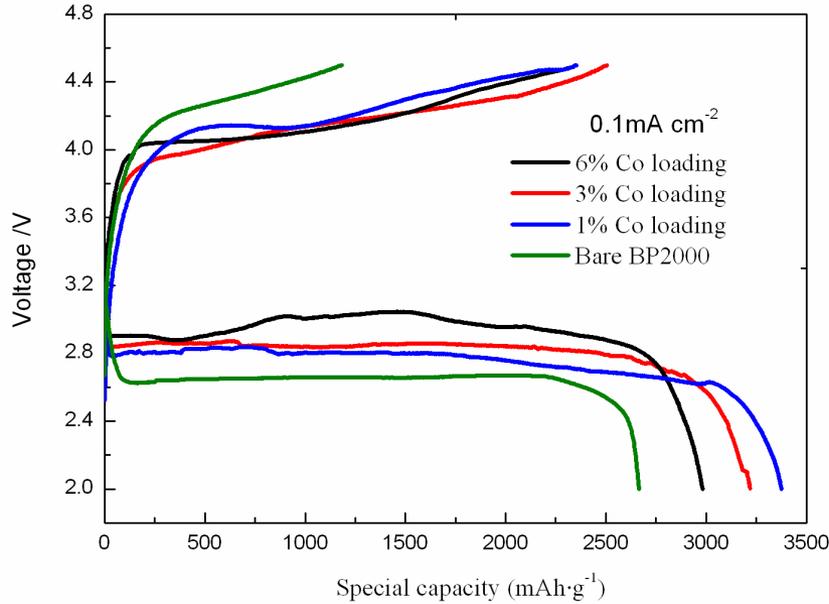
TEM a&b: Co-N/C catalyst c: BP2000



a: Co-N/C catalyst (3%Co coating)

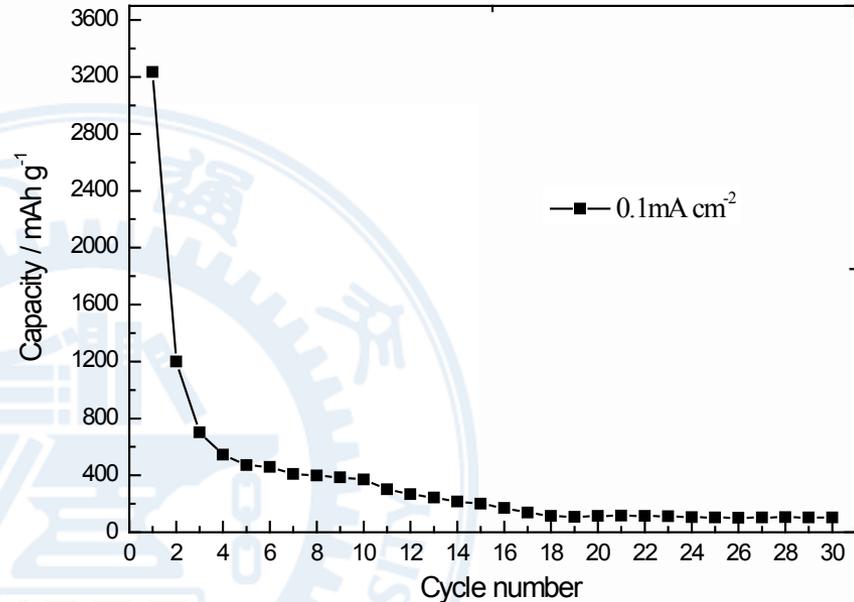
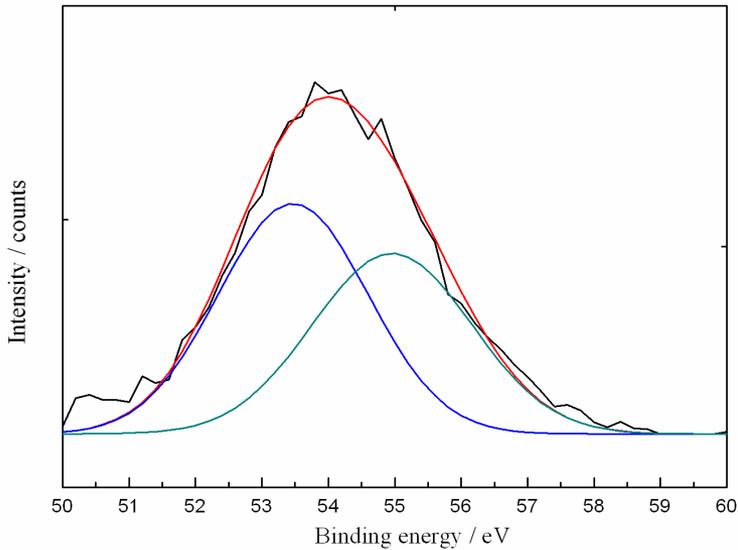
b: BP2000

	BET surface area	Pore volume
BP2000	1491m <sup>2</sup> /g	2.1cm <sup>3</sup> /g
1%Co coating	1276m <sup>2</sup> /g	2.0cm <sup>3</sup> /g
3%Co coating	864m <sup>2</sup> /g	1.8cm <sup>3</sup> /g
6%Co coating	396m <sup>2</sup> /g	1.1cm <sup>3</sup> /g



**Higher capacity (based on catalysts)**  
**Lower charge potential which tested**  
**in 1 atm oxygen atmosphere**

**3%Co loading with different current densities**



XPS spectra of Li1s was analyzed from oxygen electrode of disassembled cell after cycles. The Li 1s signal can be deconvoluted into two components. The component at 55.2 eV can be attributed to lithium peroxide. The another signal at 54 eV can be attributed to Li<sub>2</sub>CO<sub>3</sub>.

- The poor cyclability of the Li/O<sub>2</sub> cell could be attributed to the decomposition of PC-based solvent.

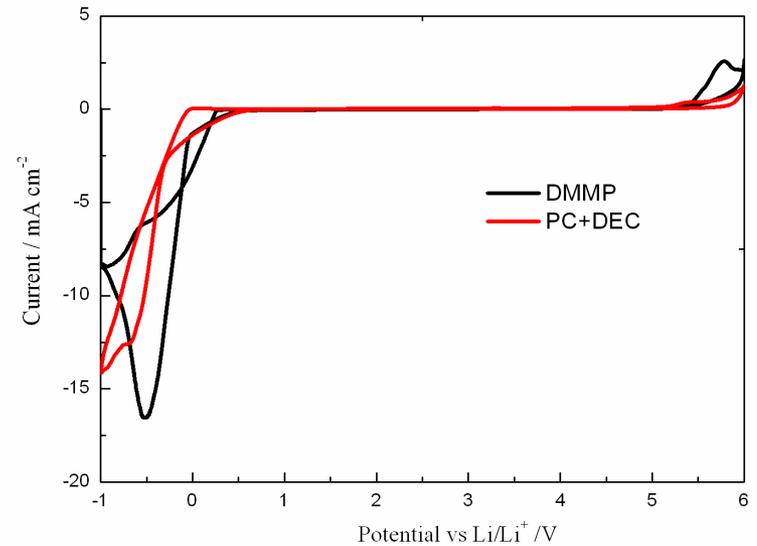
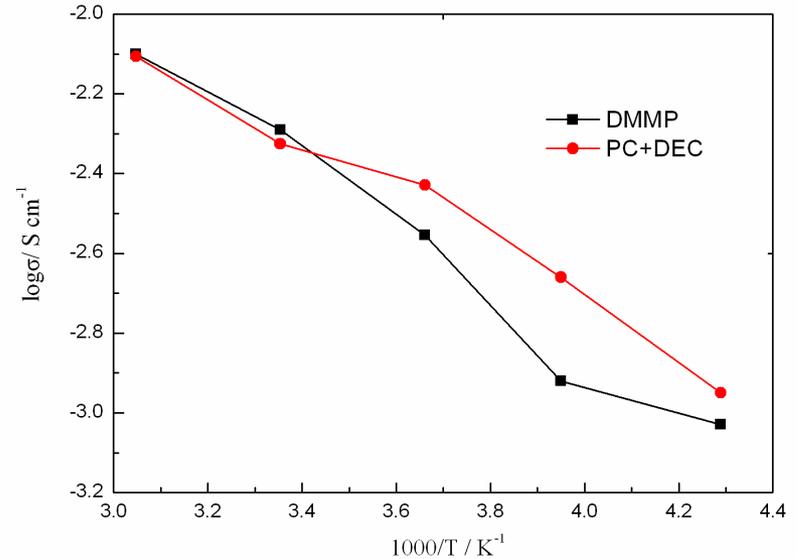
1M LITFSI in propylene carbonate (PC) and diethyl carbonate (DEC), 1:1 in volume



## 1.0 M LiTFSI-DMMP based electrolyte

1.0 M LiTFSI-DMMP solution displayed high ionic conductivity of  $\sim 5.12 \text{ mS cm}^{-1}$  at  $25^\circ\text{C}$

wide electrochemical stability windows of the 1.0 M LiTFSI-DMMP electrolyte as well as that of 1.0 M LiTFSI-PC+DEC electrolyte.





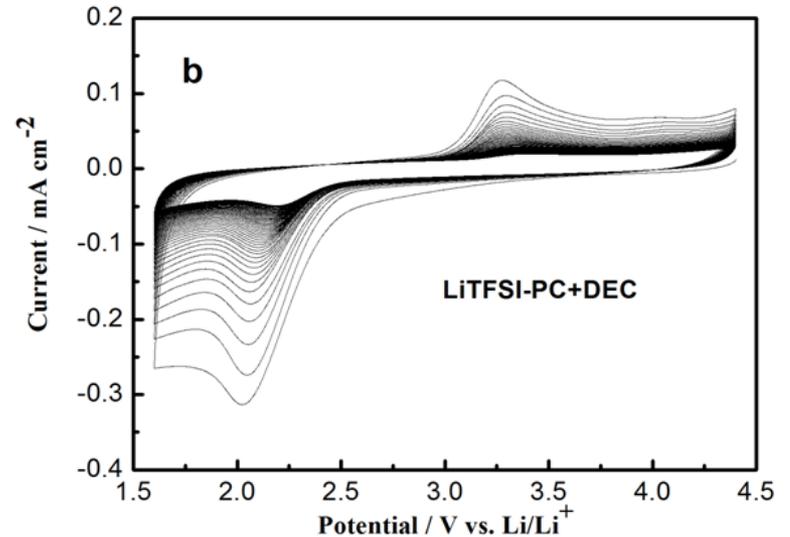
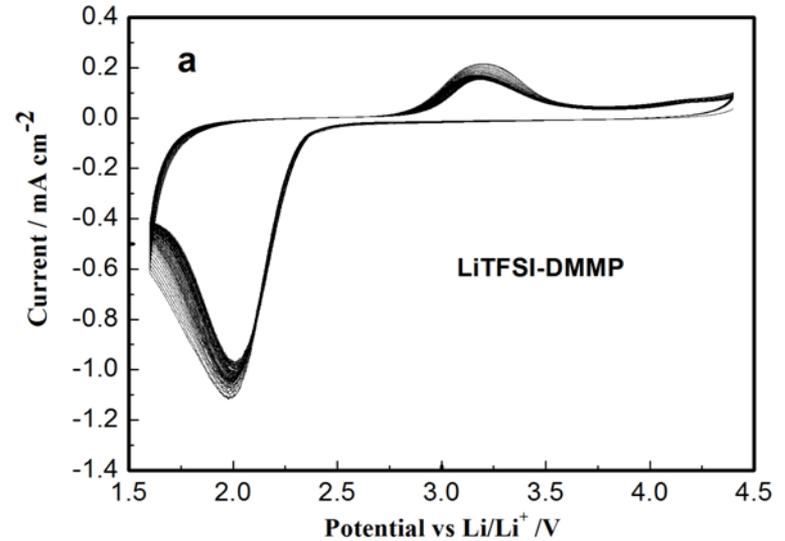
The CV test was performed under O<sub>2</sub> atmosphere (O<sub>2</sub> purged) at 100 mV s<sup>-1</sup> up to 50 cycles

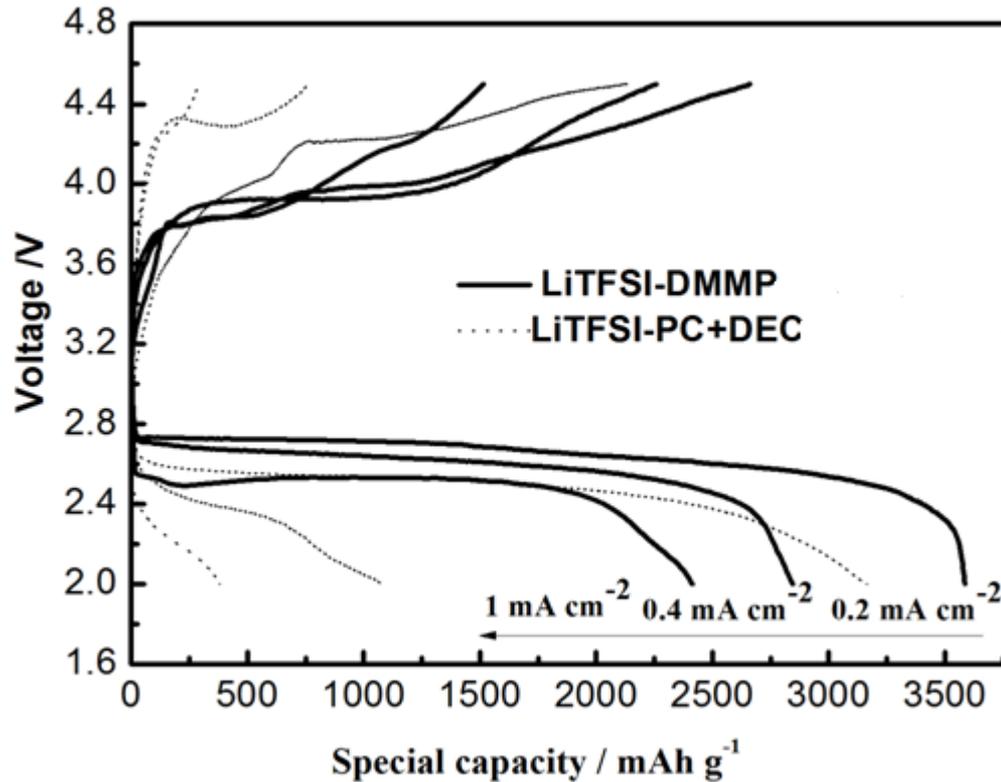
a: LiTFSI-DMMP electrolyte

b: LiTFSI-PC/DEC electrolyte

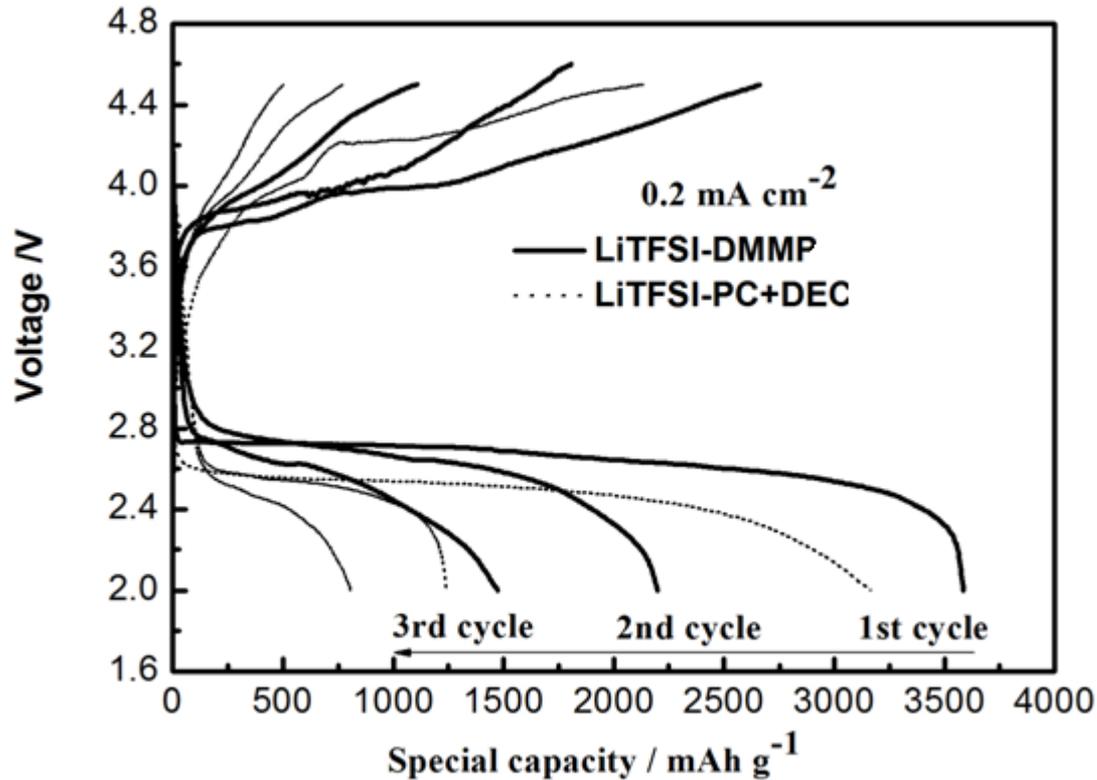
Higher reduction currents were observed in DMMP-based electrolyte than those in PC-DEC based electrolytes, which is probably due to a better O<sub>2</sub> solubility and diffusion in the DMMP-based solution.

Oxidation current peak appeared at ~3.2 V<sub>Li</sub> in DMMP-based electrolyte, while the other showed oxidation peaks at ~3.28 V<sub>Li</sub>.

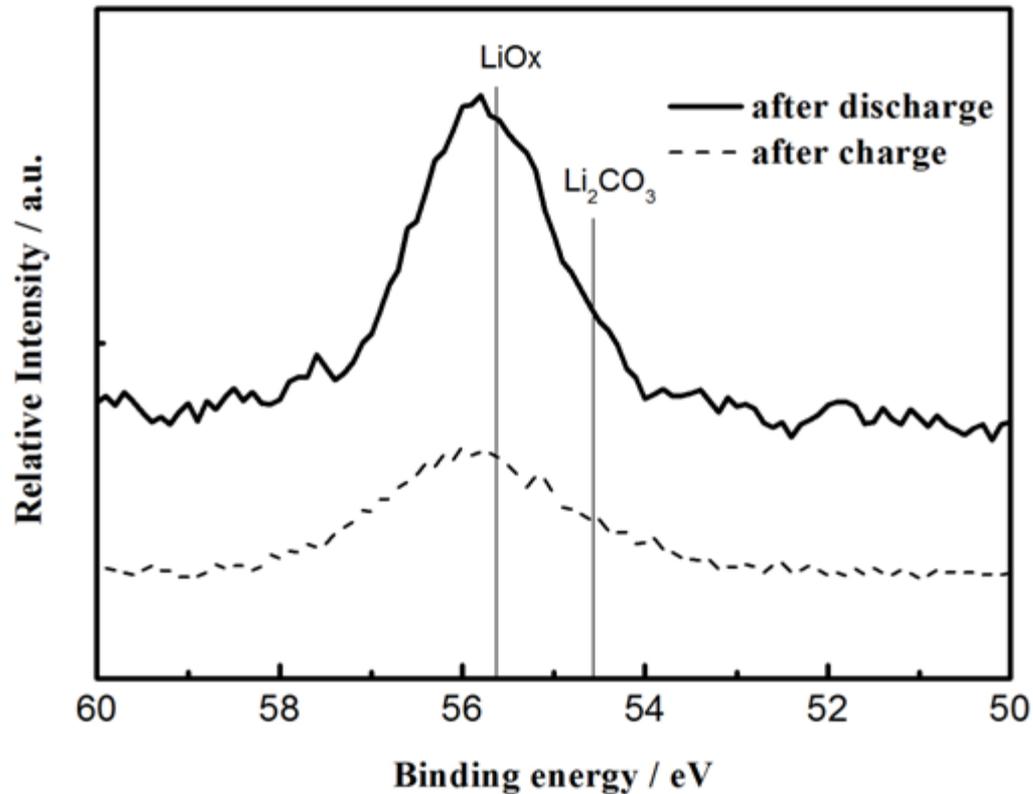




The cell exhibited reasonably high capacities at rates (0.2, 0.4 and 1 mA cm<sup>-2</sup>) with flat discharge potential plateaus. The capacities were 3585 mAh g<sup>-1</sup> (0.2 mA cm<sup>-2</sup>, average voltage plateau of 2.70 V), 2845 mAh g<sup>-1</sup> (0.4 mA cm<sup>-2</sup>, average voltage plateau of 2.65 V) and 2413 mAh g<sup>-1</sup> (1 mA cm<sup>-2</sup>, average voltage plateau of 2.53 V), respectively. The high capacities of the Li/O<sub>2</sub> cell can be attributed to the good O<sub>2</sub> solubility and diffusion in the DMMP-based electrolyte



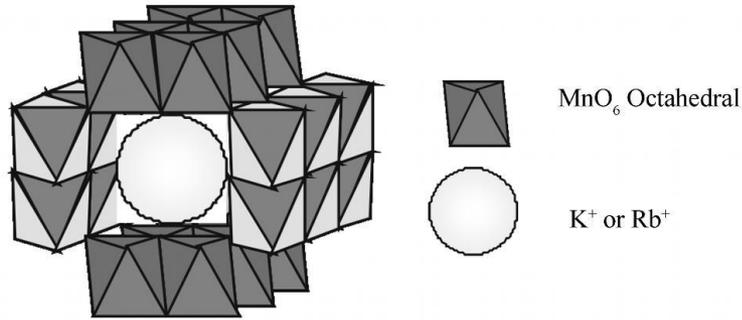
The first three cycle performance of the Li/O<sub>2</sub> cells at 0.2 mA cm<sup>-2</sup>. The cell with DMMP delivered capacities of 3585 mAh g<sup>-1</sup> (1st cycle), 2199 mAh g<sup>-1</sup> (2nd cycle) and 1472 mAh g<sup>-1</sup> (3rd cycle), while the another cell with PC electrolyte showed capacities of 3154 mAh g<sup>-1</sup> (1st cycle), 1241 mAh g<sup>-1</sup> (2nd cycle) and 807 mAh g<sup>-1</sup> (3rd cycle).



The Li 1s XPS spectra of the product samples the major oxygen reduction products in DMMP-based electrolyte are lithium oxides. The XPS spectrum of the charged sample shows that the reduction product can not be completely decomposed during charge.



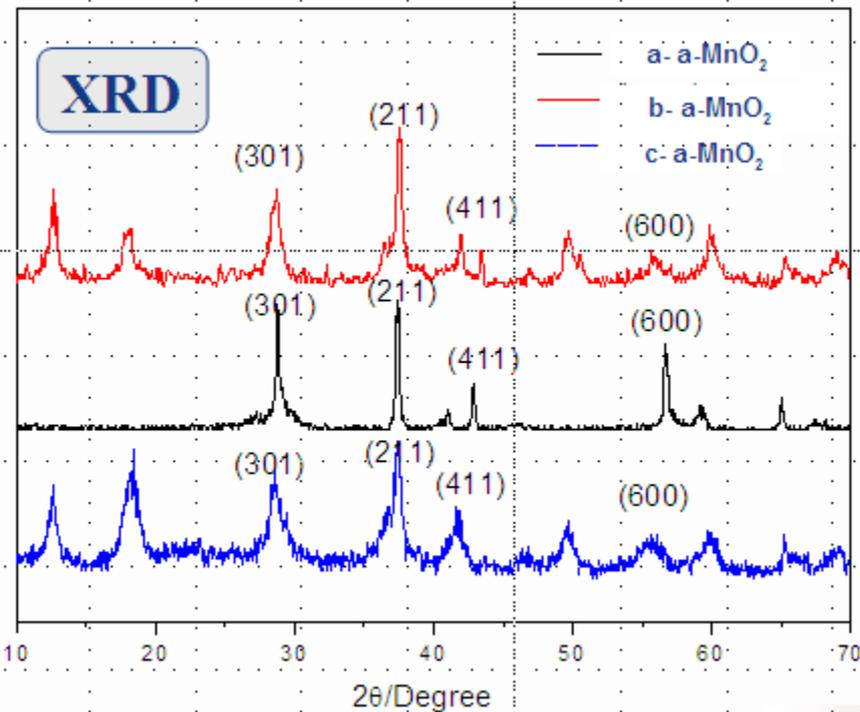
# $MnO_x$ based catalysts



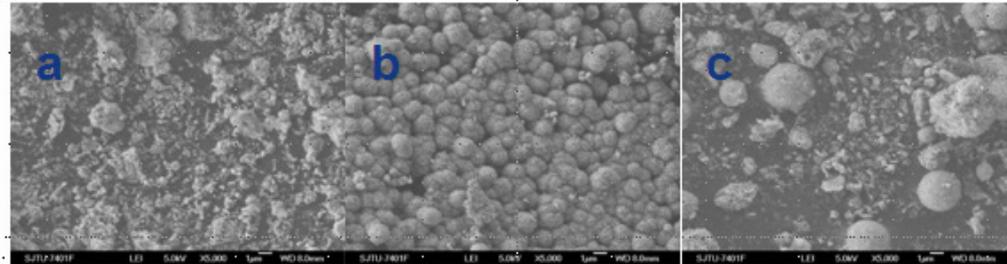
## Why a- $MnO_2$ ?

1. The structure of a- $MnO_2$  ([2x2], or [1x1]) contains large tunnels or holes, which help in the transfer of the products after discharge/charge
2. low cost, and easy to be prepared

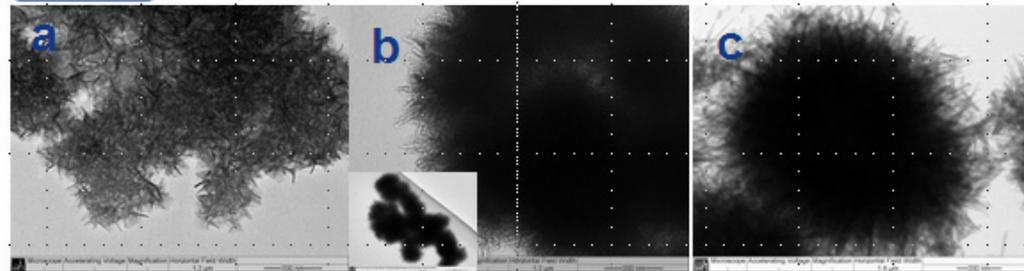
XRD



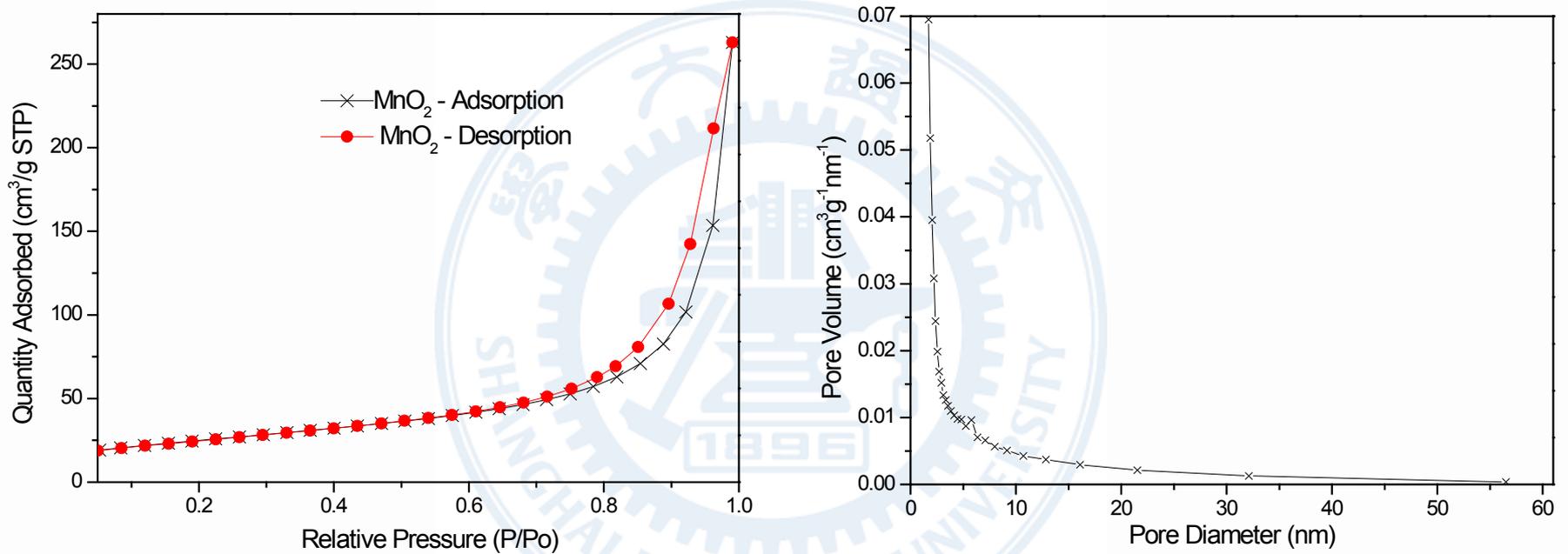
SEM



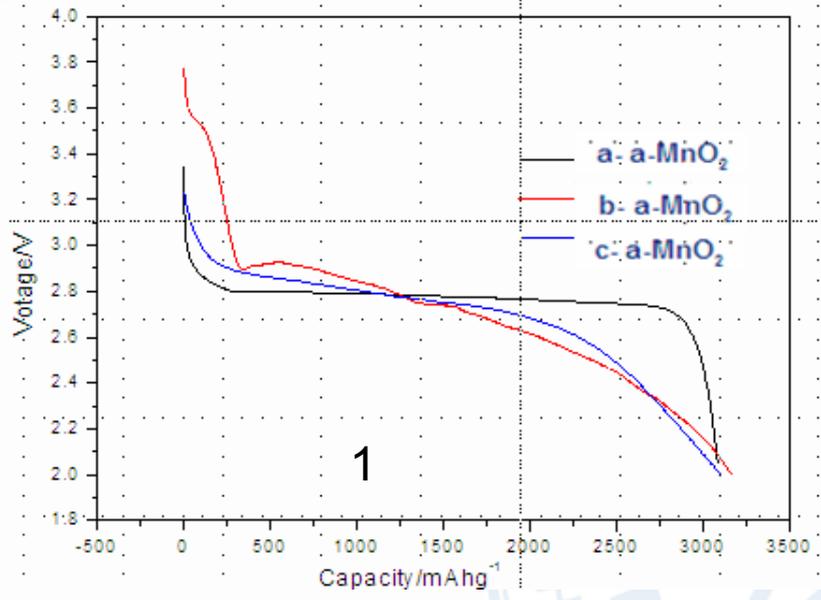
TEM



SEM and TEM of different a- $MnO_2$  prepared in our experiment



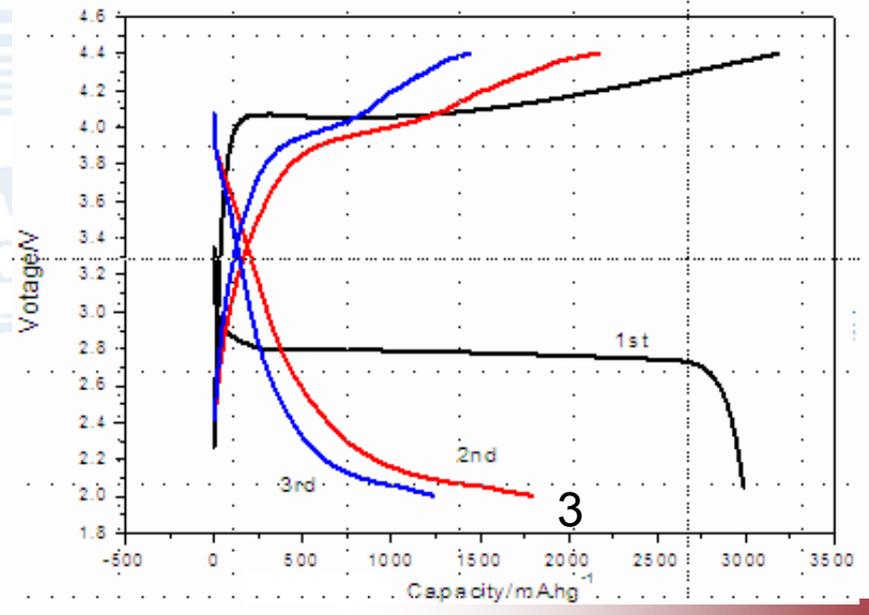
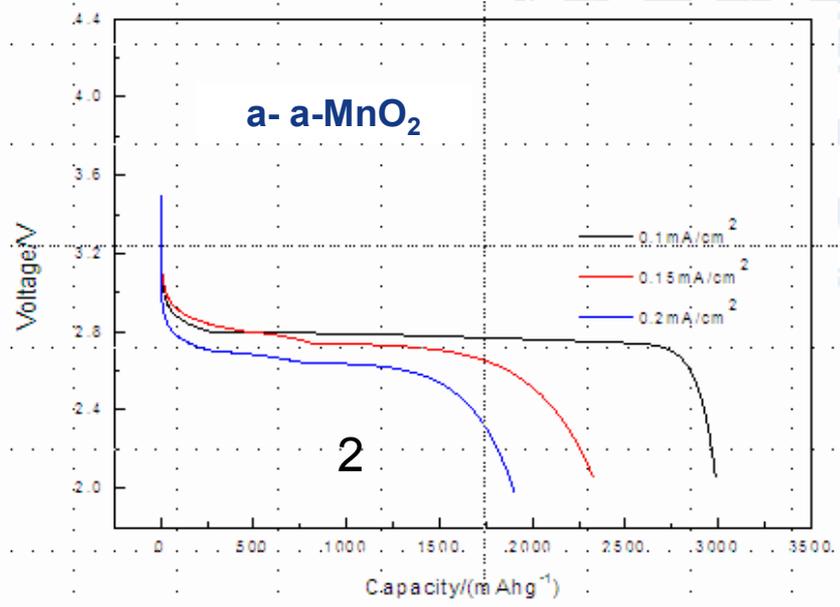
**The specific surface area (BET) and pore size and pore distribution of a-MnO<sub>2</sub> nanowires catalyst**



1: Discharge curves of different a-MnO<sub>2</sub> under current 0.1 mA/cm<sup>2</sup>

2: Discharge curves of type a a-MnO<sub>2</sub> under current 0.1, 0.15, 0.2 mA/cm<sup>2</sup>, respectively

3: Cycle performance of type a a-MnO<sub>2</sub> under current 0.1 mA/cm<sup>2</sup>





1. Improve the Rate performance
2. Improve the discharge voltage

and reduce the charging voltage

Poor Cycle performance

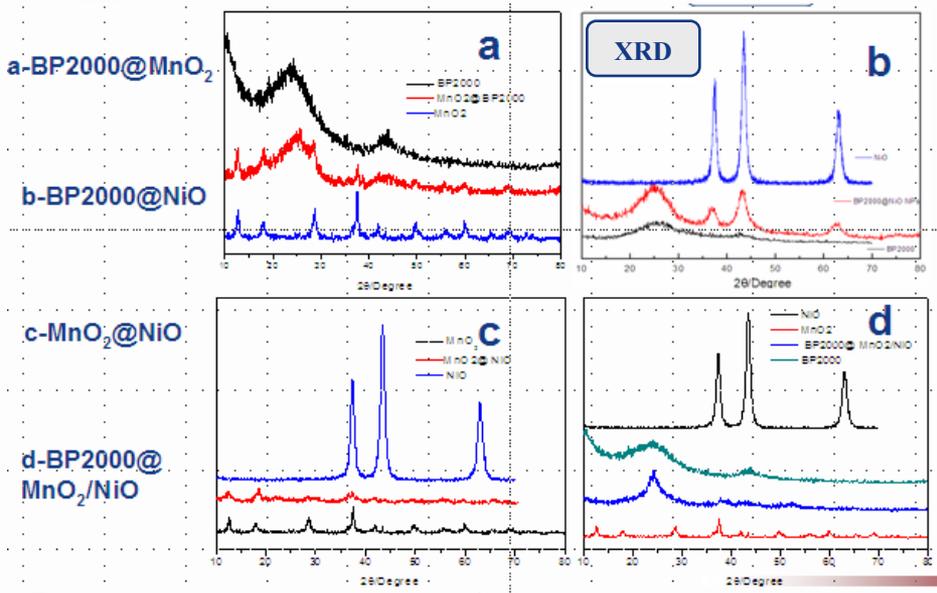
Cocatalyst



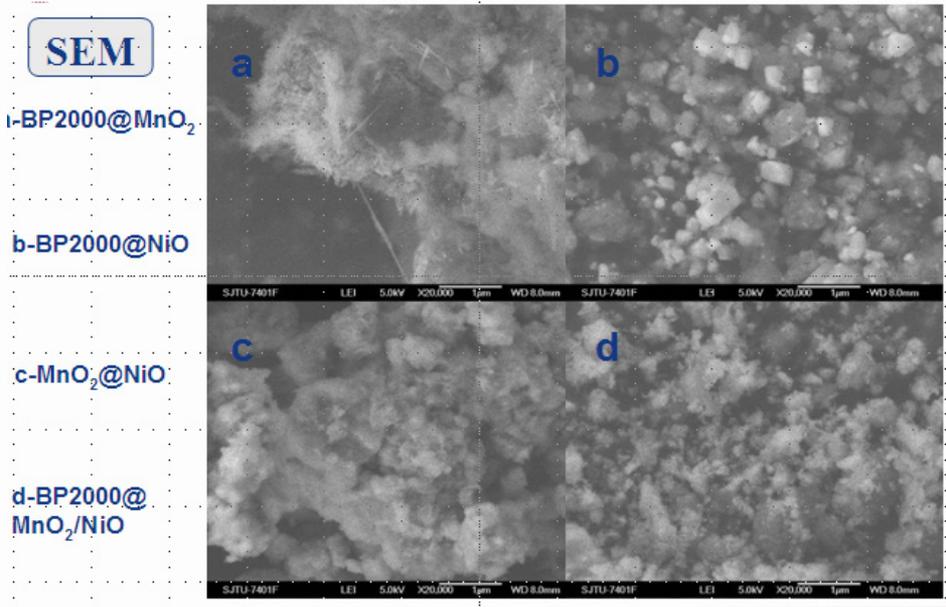


## Why MnO<sub>2</sub>-based catalyst?

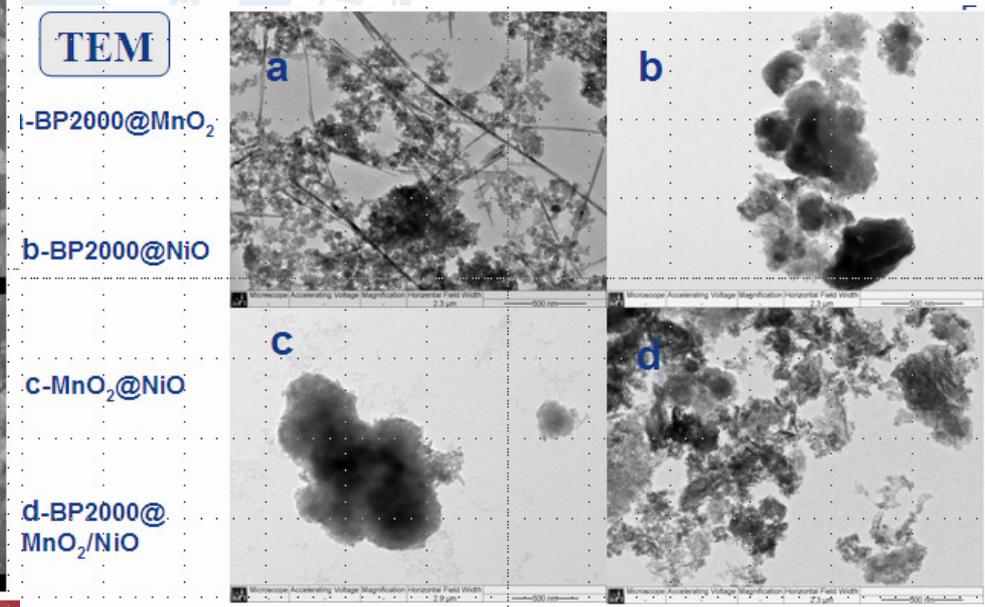
1. Insulation material, apt to increase the Lithium-air battery's internal resistance
2. Cocatalyst, such as NiO, CeO<sub>2</sub>, can effectively improve the Catalytic properties.

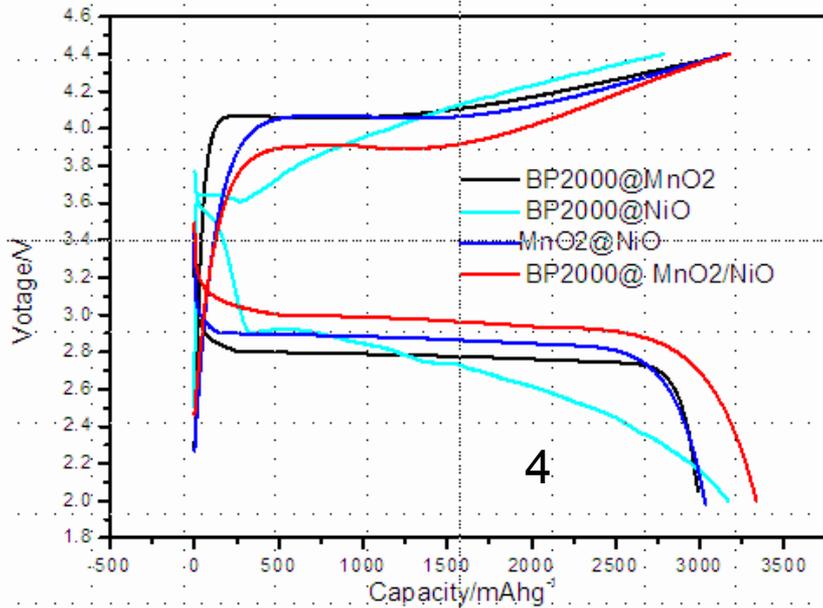


### SEM



### TEM

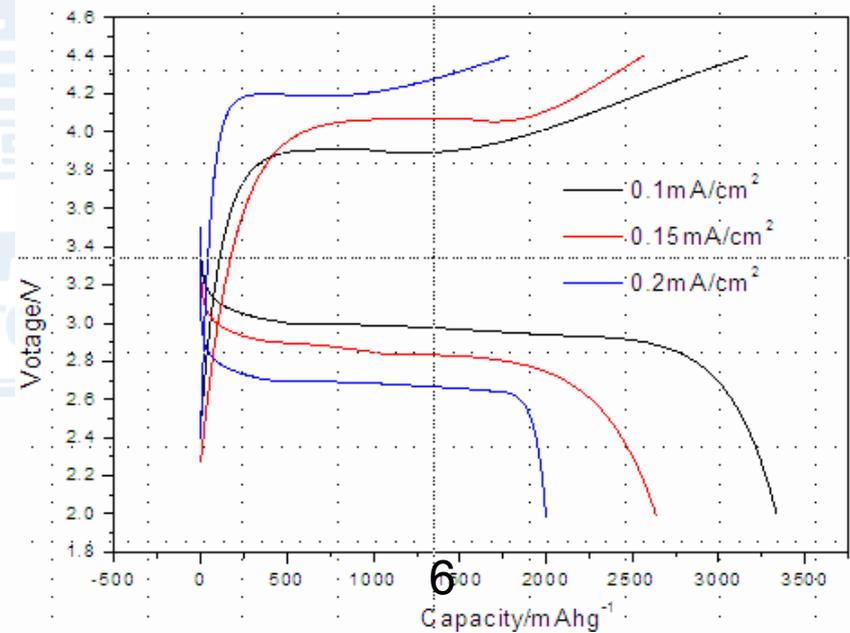
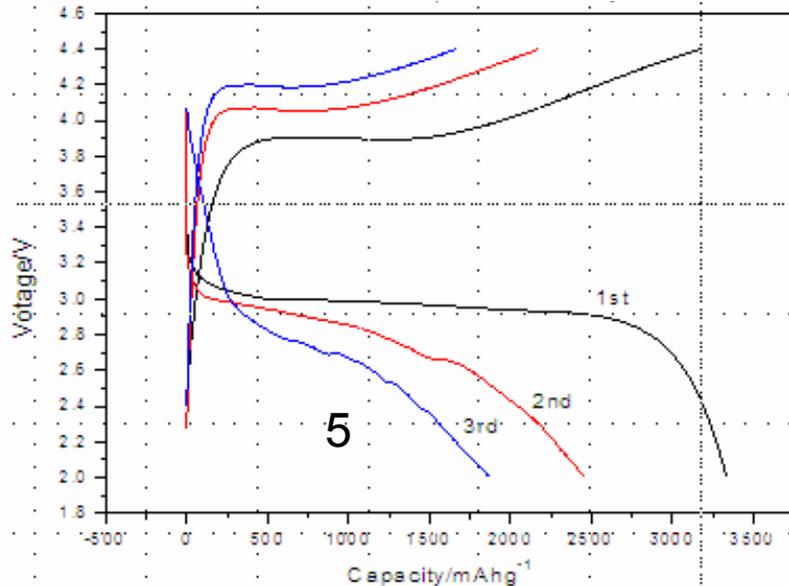




4: Discharge curves of different a-MnO<sub>2</sub> under current 0.1 mA/cm<sup>2</sup>

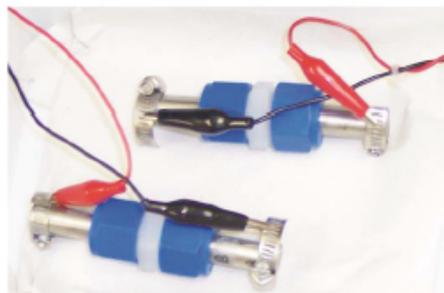
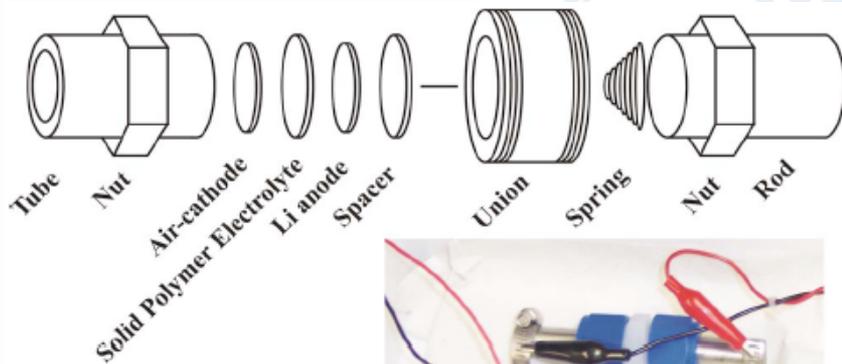
5: Discharge curves of type a a-MnO<sub>2</sub> under current 0.1, 0.15, 0.2 mA/cm<sup>2</sup>, respectively

6: Cycle performance of type a a-MnO<sub>2</sub> under current 0.1 mA/cm<sup>2</sup>

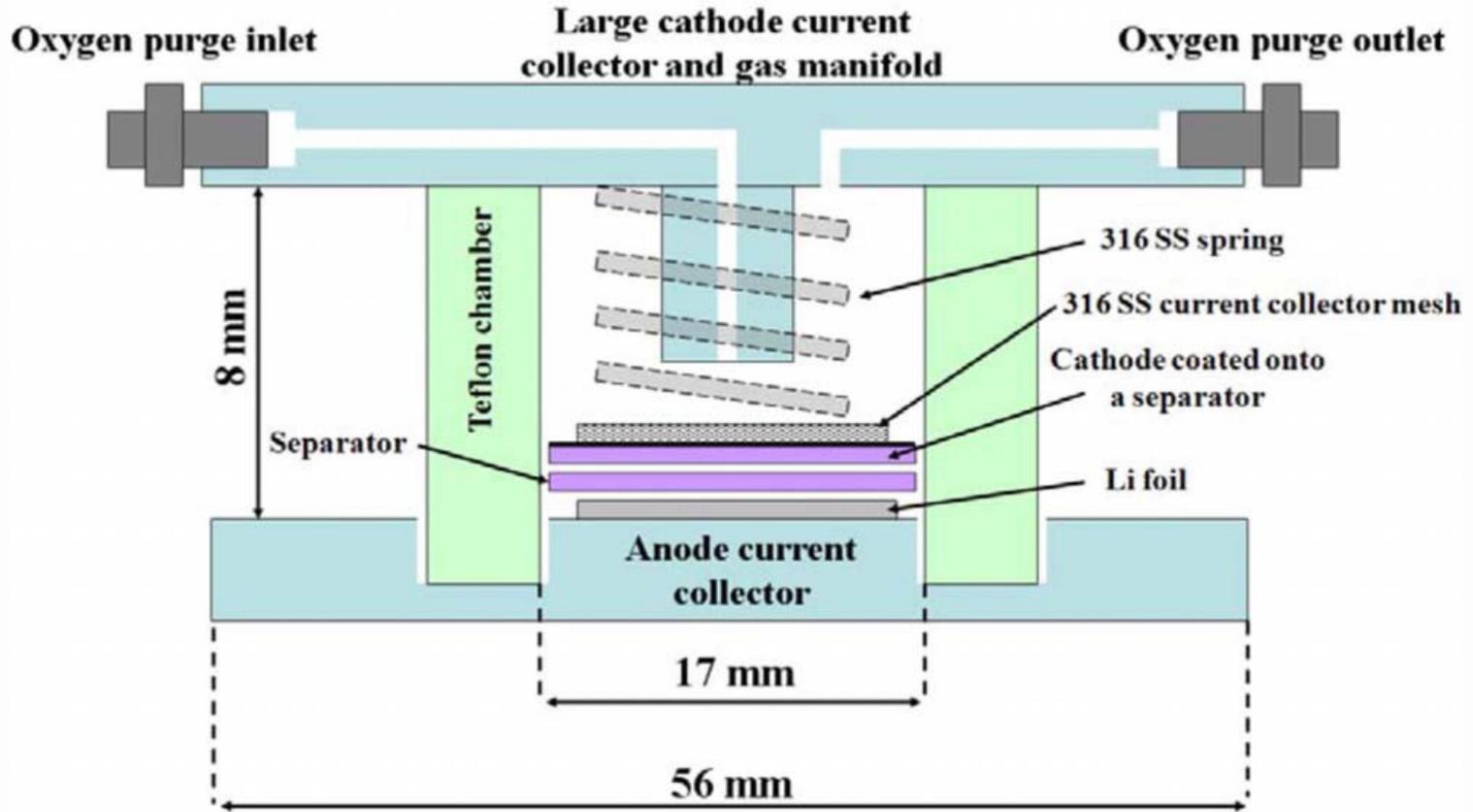




# Battery Module

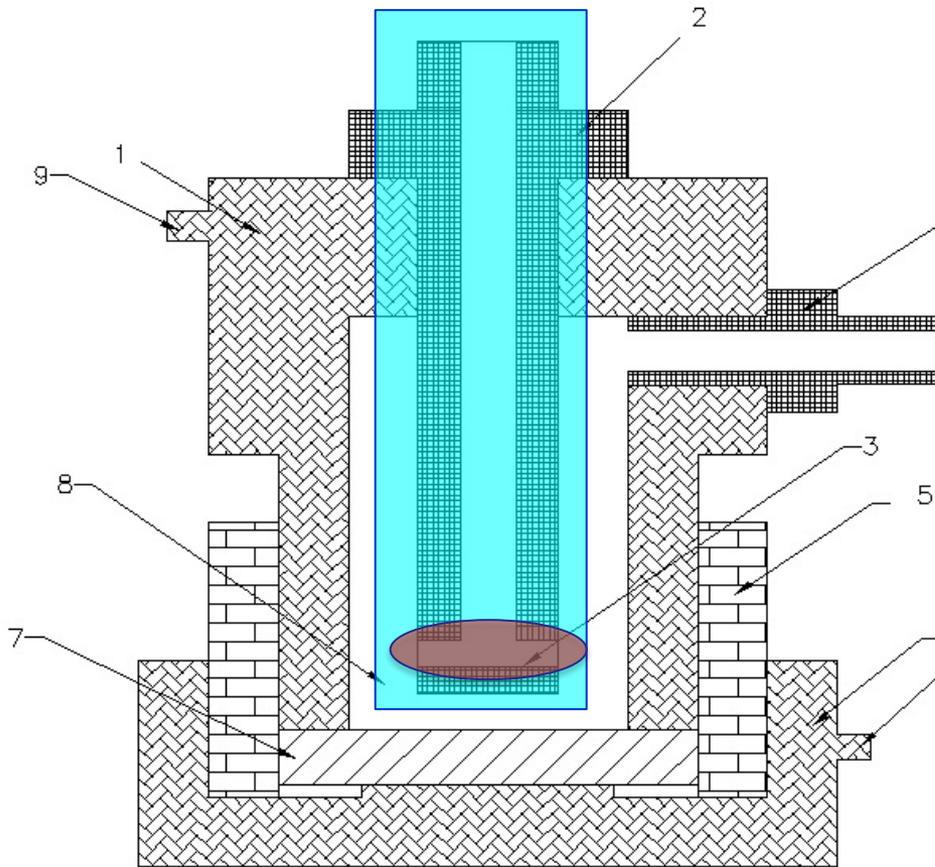


**Swagelok Cell**

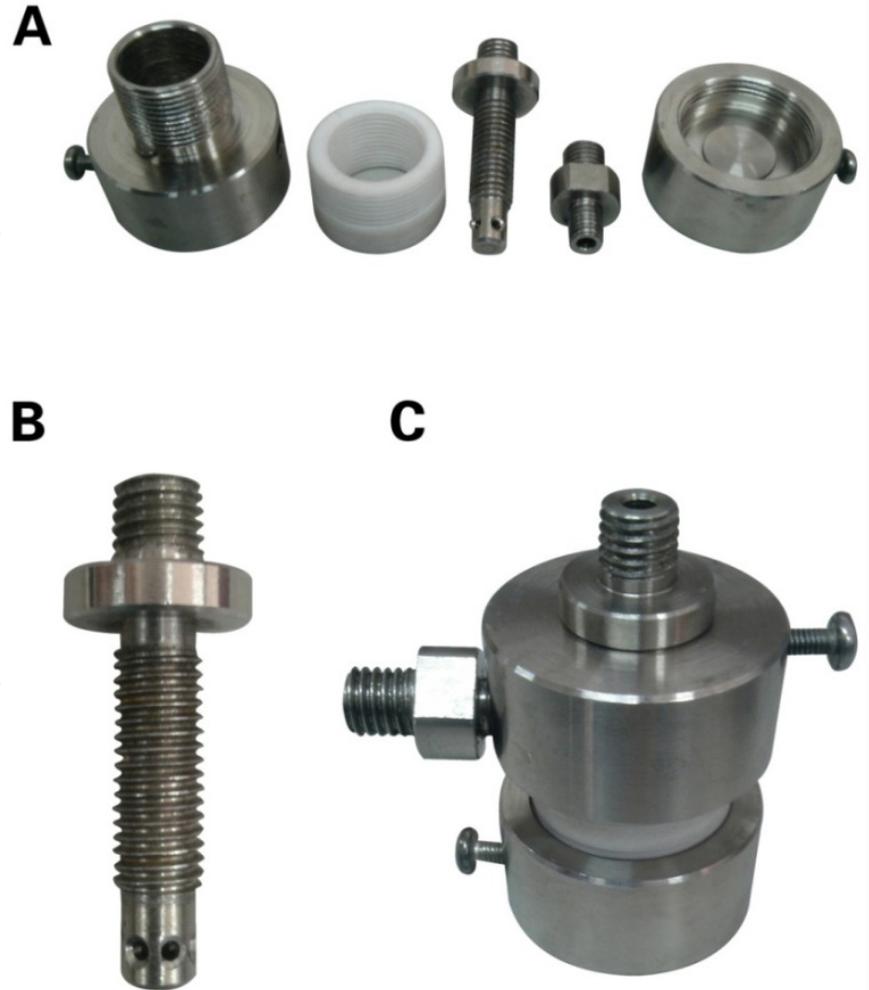


## Yang Shao-Horn's Cell

Electrochemical and Solid-State Letters, 13(2010)A69-A72

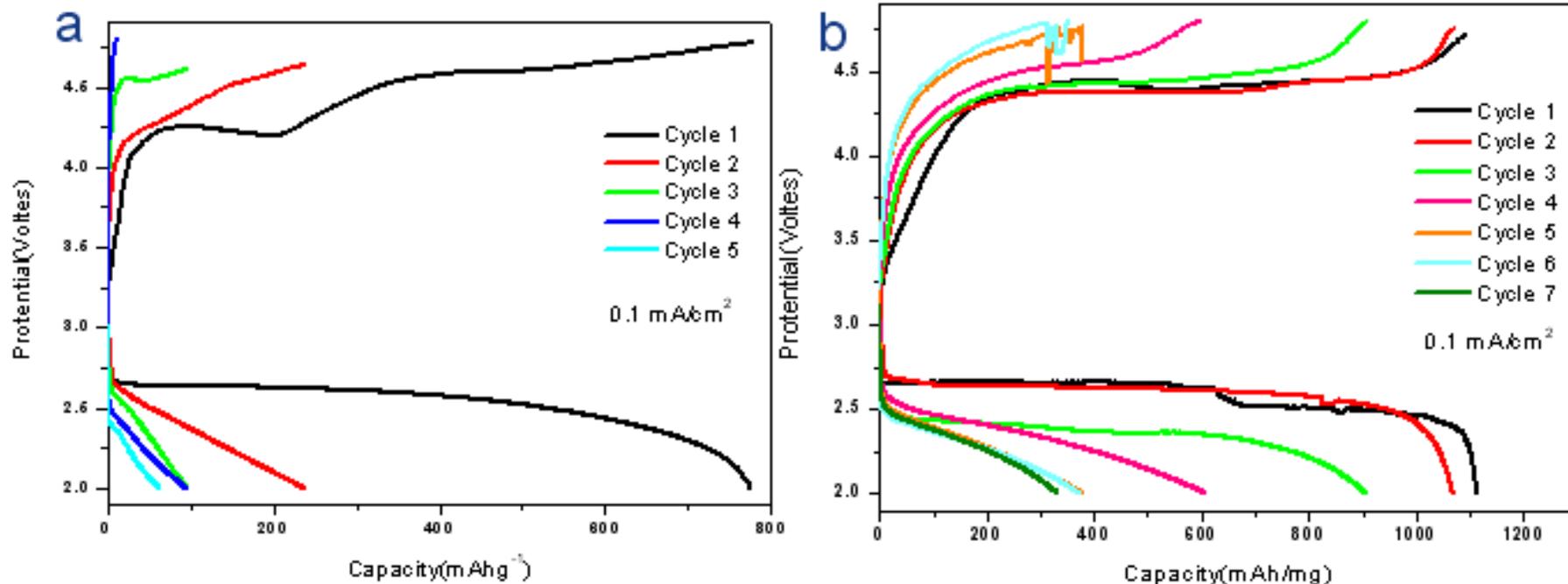


**SJTU Cell**

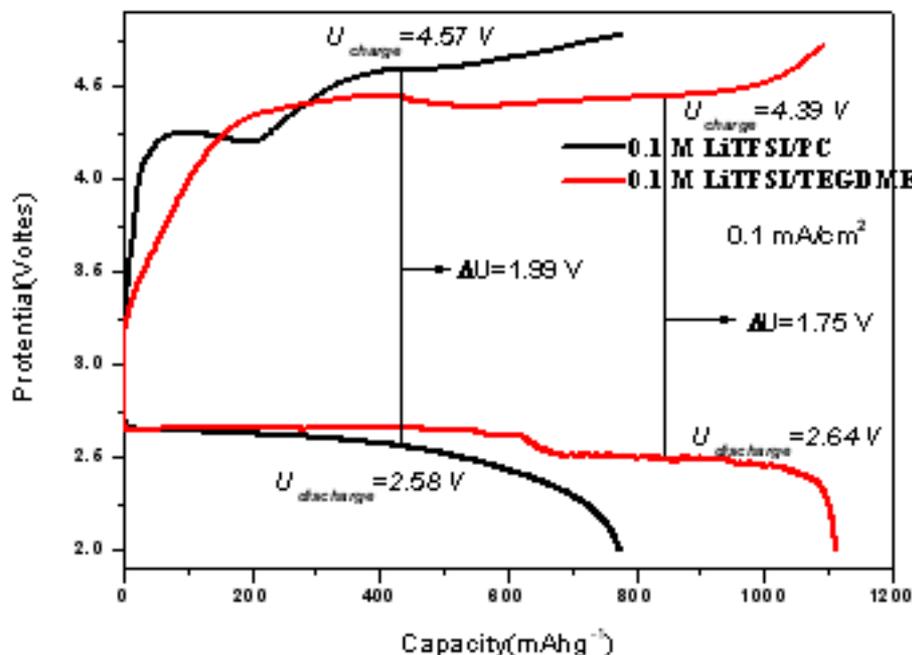


# Battery Module

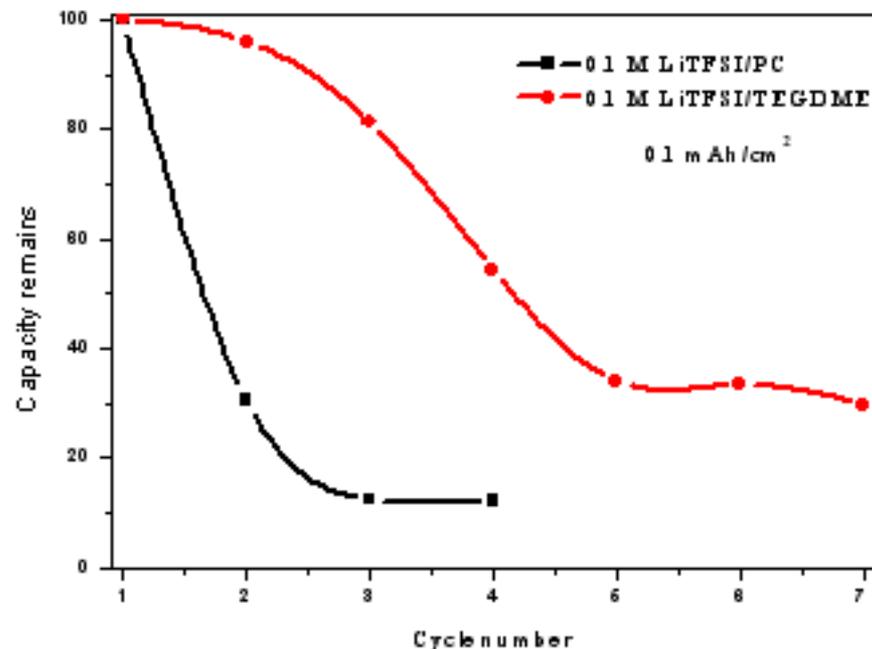
## Effect of electrolyte on electrochemical performance



Discharge/charge curves using 0.1 M LiTFSI/ PC (a) and TEGDME (b) Li-O<sub>2</sub> cells at 0.1 mA cm<sup>-2</sup> in oxygen

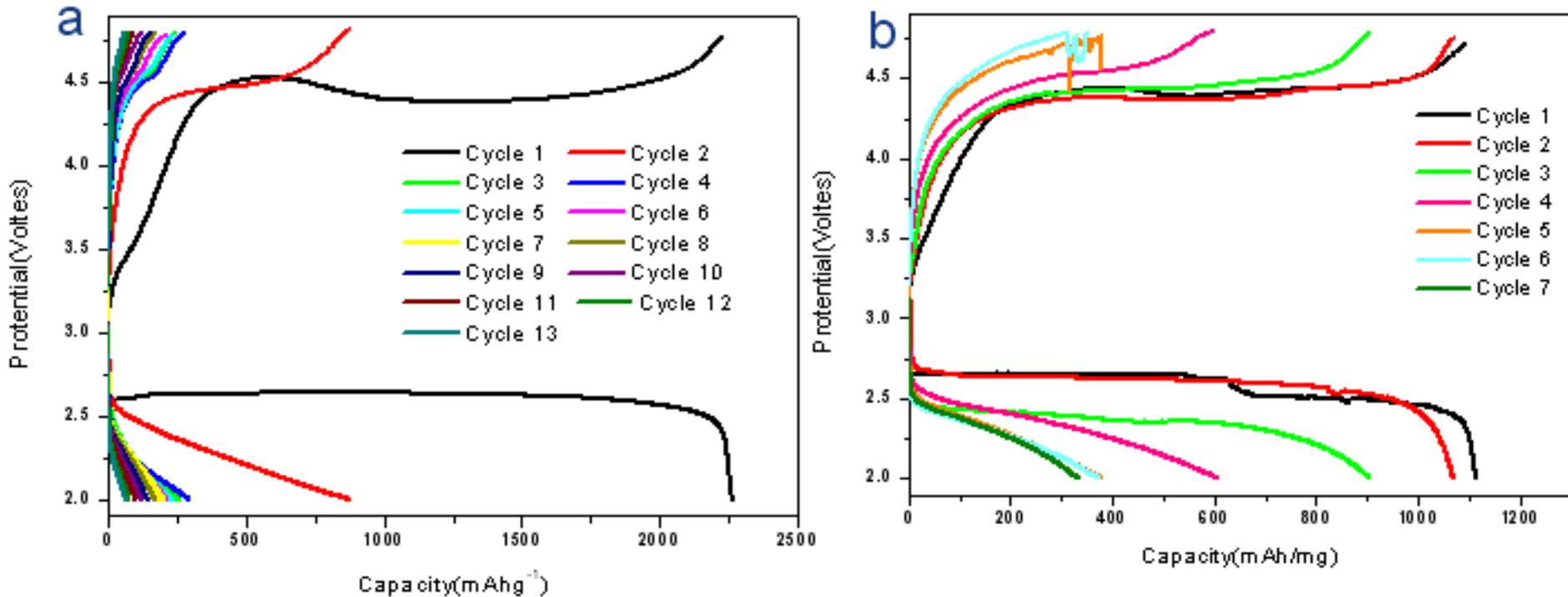


First discharge/charge curves of Li-O<sub>2</sub> cells using the 0.1 M LiTFSI/ PC and TEGDME as electrolyte at 0.1 mA cm<sup>-2</sup> in oxygen

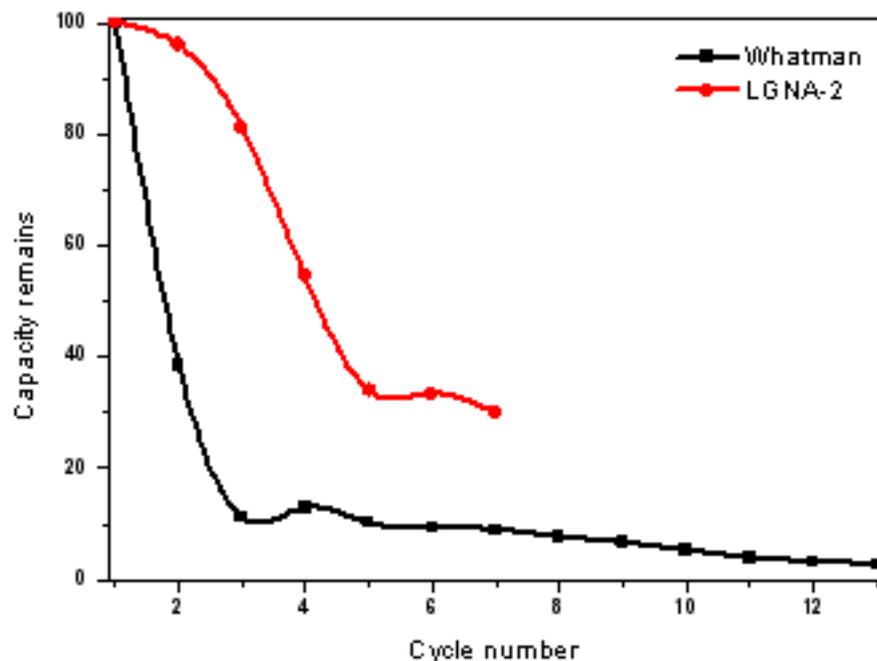
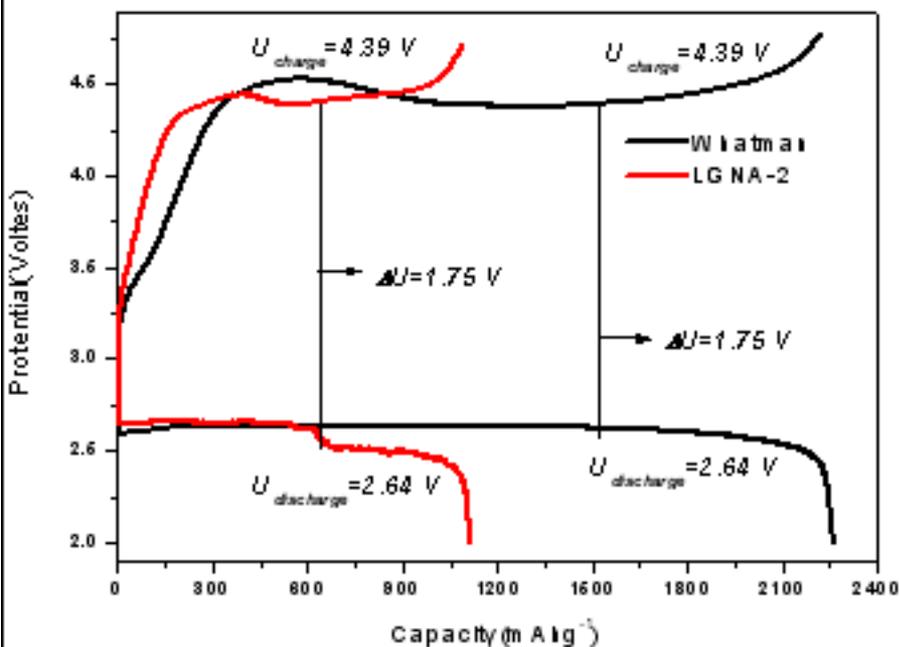


Discharge capacities remains versus cycle numbers using the 0.1 M LiTFSI/ PC and TEGDME as electrolyte at 0.1 mA cm<sup>-2</sup> in oxygen

## Effect of separator on electrochemical performance



Discharge/charge curves using Whatman (a) and LGNA-2 (b) Li-O<sub>2</sub> cells at 0.1 mA cm<sup>-2</sup> in oxygen

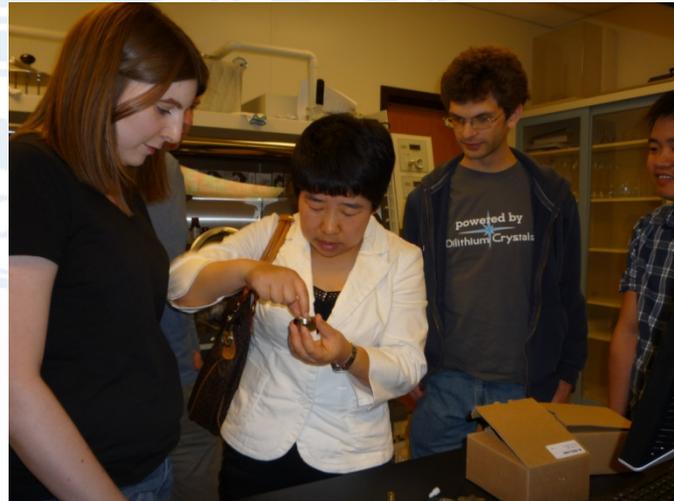


First discharge/charge curves of Li-O<sub>2</sub> cells using the Whatman and LGNA-2 as membrane at 0.1 mA cm<sup>-2</sup> in oxygen

Discharge capacities remains versus cycle numbers using the Whatman and LGNA-2 as membrane at 0.1 mA cm<sup>-2</sup> in oxygen



- ◆ Prof Ma attended the U.S.–China Electric Vehicle and Battery Technology Workshop, August 1-6, 2011, Chicago, USA
- ◆ Prof Ma will attend the U.S.–China Electric Vehicle and Battery Technology Workshop on August 23-24, 2012, Boston, USA
- ◆ Prof Xianxia Yuan visited UM on May 9-12, 2012
- ◆ Dr. Max Radin and Dr. Feng Tian in Donald J. Siegel's group visited SJTU on October 17 to 26, 2011





## Proposed Timeline for Major Tasks and Milestones

### UM/Siegel:

- A. (0-9 months) Evaluate concentrations and diffusivities of intrinsic defects using “bulk morphology” of catalytic decomposition of  $\text{Li}_2\text{O}_2$ .
- B. (9-12 months) Dependence on temperature and pressure.
- C. (12-21 months) Metal catalyst screening.
- D. (21-24 months) Preliminary investigation of “interface morphology” of catalytic decomposition.

### SJTU/Ma-Yuan:

- E. (0-12 months) Design, preparation, and characterization of catalyst structure and performance. Activation energy of  $\text{Li}_2\text{O}_2$  decomposition.
- F. (12-18 months) Comparative study of experimental results and modeling analysis; Improvement and optimization of catalysts.
- G. (19-24 months) Evaluate dependence of battery performance on cathode structure.



## Follow-on funding in China:

- ◆ **RMB 600,000, NSFC: Study on microstructure and key materials of cathode in lithium-air battery (21176155) (PI: Xianxia Yuan)**
- ◆ **RMB 360,000, NSFC: Electrochemical basic study on the key materials for Li-air cell stack design (21073120) (PI: Xiaozhen Liao)**
- ◆ **RMB300,000, NSF Shanghai: Oxygen reduction reaction Catalysts preparation and its fuel cell and Li-air battery application (10JC1406900) (PI: Xianxia Yuan)**
- ◆ **RMB 8,000,000, Sinopoly Battery Research Center (SBRC,Hong Kong), Next generation battery development for electric vehicle application (2011.09-2015.08) (PI: Zi-Feng Ma)**



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- ◆ Li He, Xiao-Zhen Liao, Ke Yang, Yu-Shi He, Wen Wen, Zi-Feng Ma, Electrochemical Characteristics and Intercalation mechanism of ZnS/C Composite as Anode Active Material for Lithium-ion Batteries, *Electrochim. Acta*, 2011, 56 (3) : 1213-1218
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- ◆ Hui-Juan Zhang, Xianxia Yuan, Zi-Feng Ma, Zongping Shao. Synthesis and characterization of binary non-precious metal electrocatalyst for oxygen reduction reaction in proton exchange membrane fuel cells. Electrochimica Acta. Doi: 10.1016/j.electacta.2012.06.011.
- ◆ Hai-Chuan Kong, Xianxia Yuan, Xiao-Yun Xia, Zi-Feng Ma. Effects of preparation on electrochemical properties of CoTMPP/C as catalyst for oxygen reduction reaction in acid media. International Journal of Hydrogen Energy. DOI: 10.1016/j.ijhydene.2012.03.025.
- ◆ Xianxia Yuan, Zi-Feng Ma, et al. Properties of Co-PPy-TsOH/C prepared from various cobalt precursors as catalyst towards oxygen reduction reaction. Applied catalysis B: environmental, revised



- 1 Invention patent: Hao-Dong Sha, Xianxia Yuan, Zhong Ma, Zi-Feng Ma. A novel mould for lithium-air battery. Application number: 201210134349.5**
- 2 Invention patent: Hong Wang, Xiao-Zhen Liao, Zi-Feng Ma, A novel cathode catalyst and preparation method for lithium air battery. Application number: 201110131455.3**
- 3 Utility model patent: Zhong Ma, Xianxia Yuan, Hao-Dong Sha, Zi-Feng Ma. A novel mould for lithium-air battery. Application number: 201220197482.0**

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